

# Energy Storage material status and challenges for KSA and practical application of 3D holey-graphene structure

**Imran Shakir**



# Energy Storage Research Group



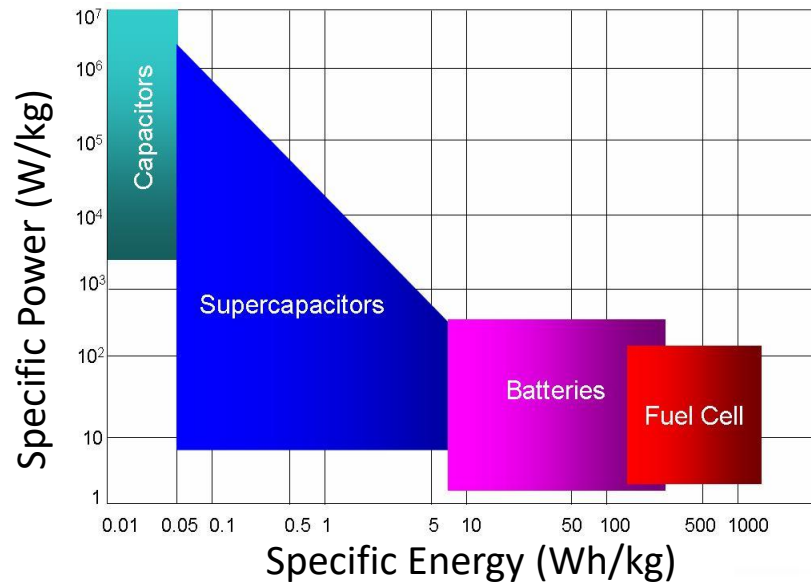
## Objective

Development of low-cost, high-performance functional materials for large-scale, affordable storage devices for portable electronics, vehicle electrification, smart grid as well as the broad-scale deployment of alternative energy resources such as solar and wind.

## Portable electronics



20-30% CO<sub>2</sub> emission



Energy storage is one of the key challenges we face in the 21<sup>st</sup> century



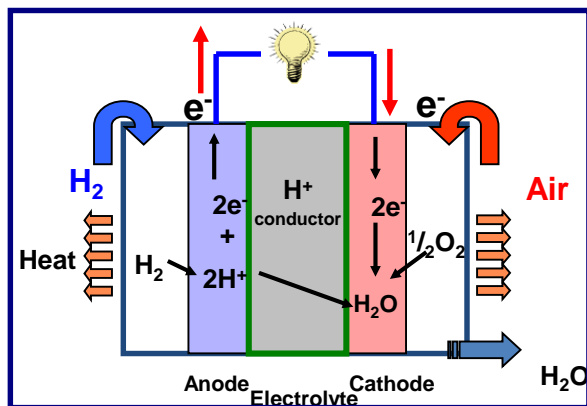
# ELECTROCHEMICAL ENERGY TECHNOLOGIES



## Alternative Energy Technologies

- Solar, wind, nuclear, hydro, geothermal, fuel cells, batteries, supercapacitors
- **Fuel cells, batteries, supercapacitors:** Only viable option for automobiles (~ 30%)
- **Batteries:** Critical for storing and efficiently utilizing solar and wind energies

### Fuel Cell

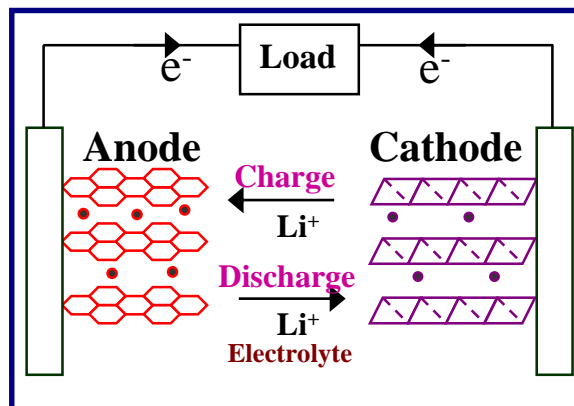


### Conversion Device

Portable, transportation, & stationary



### Battery

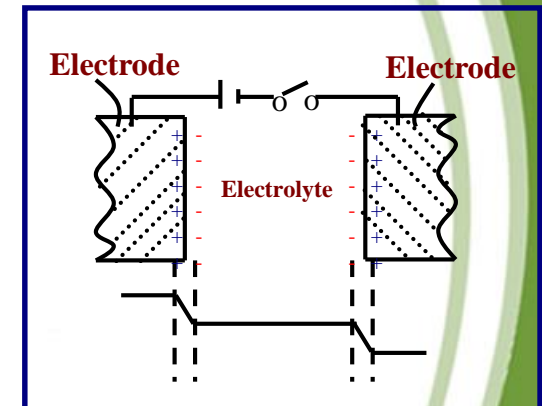


### Storage Device

Portable, transportation, & stationary



### Supercapacitor



### Storage Device

Portable & transportation

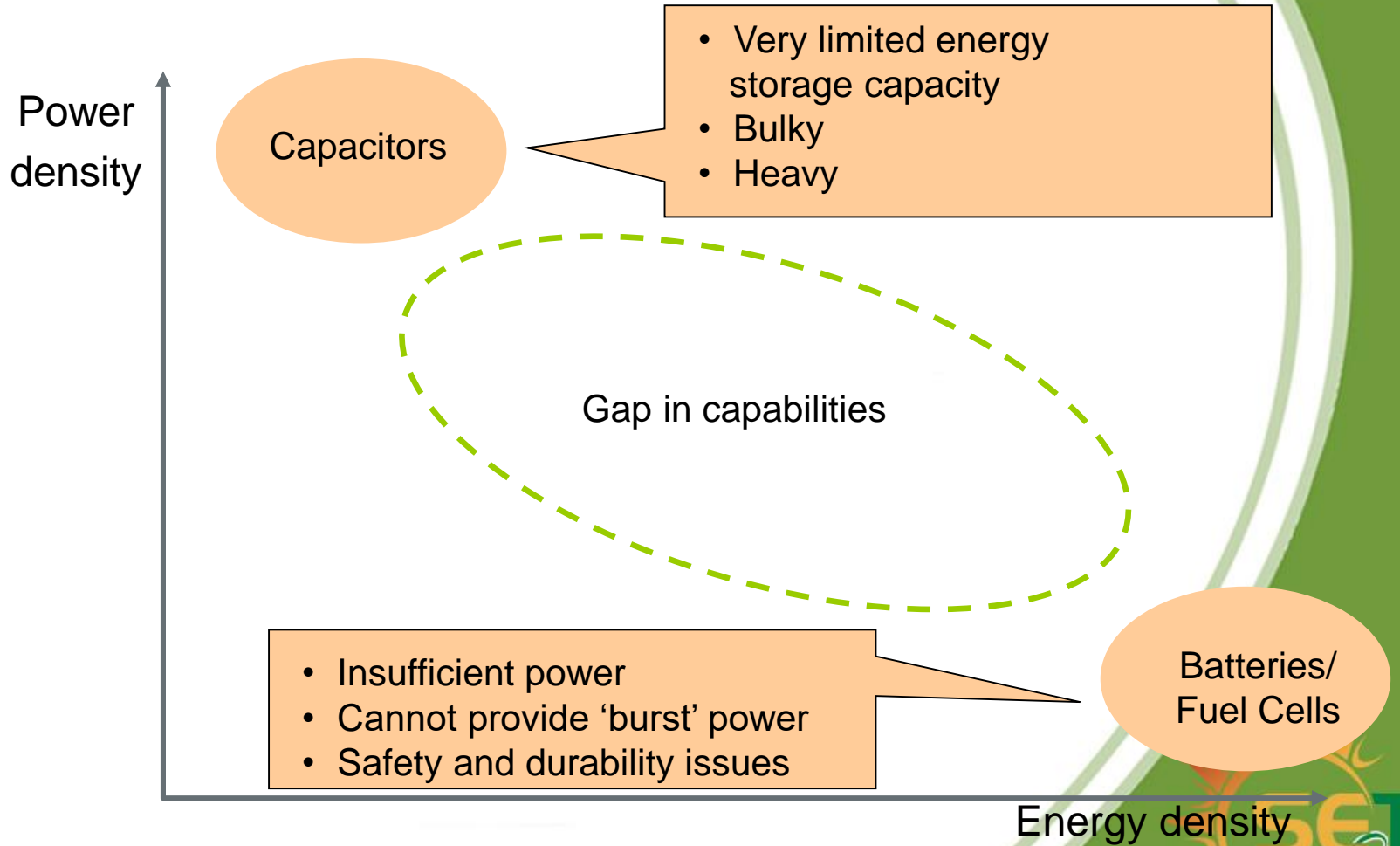
### On-Site Power Generation



- Chemical energy directly into electrical energy – clean energy technologies
- **Challenges:** high cost, safety, durability, & operability problems

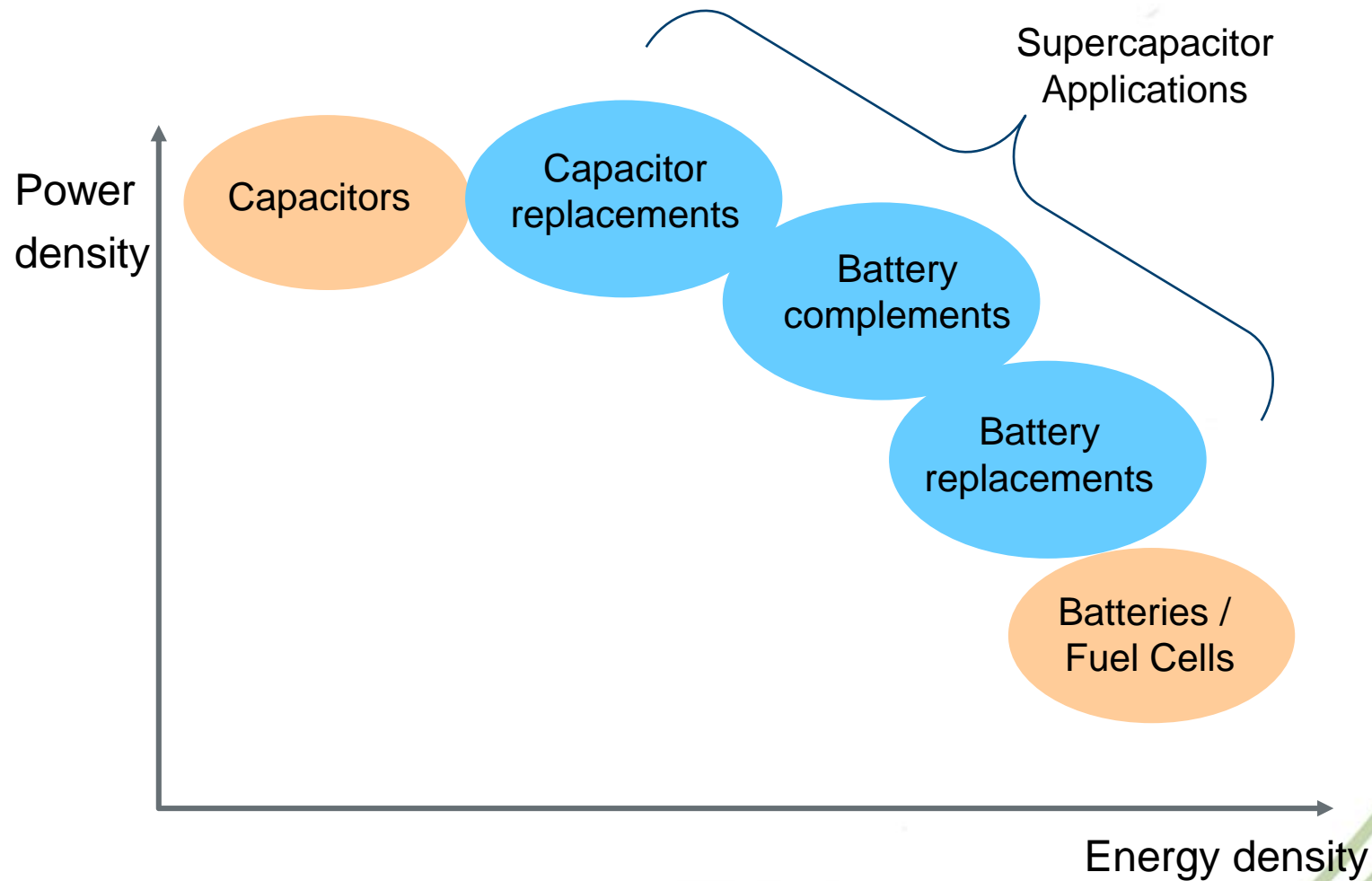


# Current Energy Storage Devices



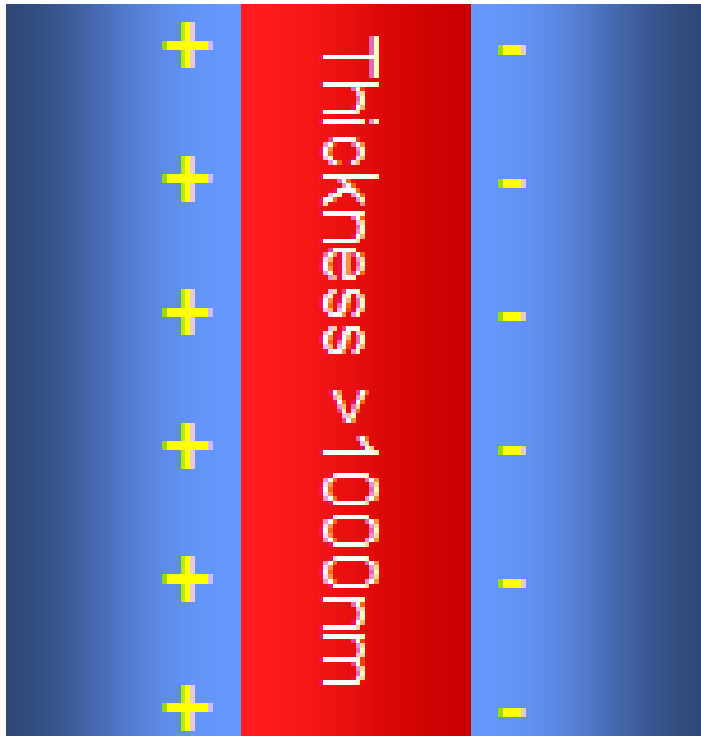
# Current Energy Storage Devices

*Supercapacitors have a unique ability to provide a solution that is small, lightweight and has the power to fill the gap in capabilities*



# Energy Storage Devices

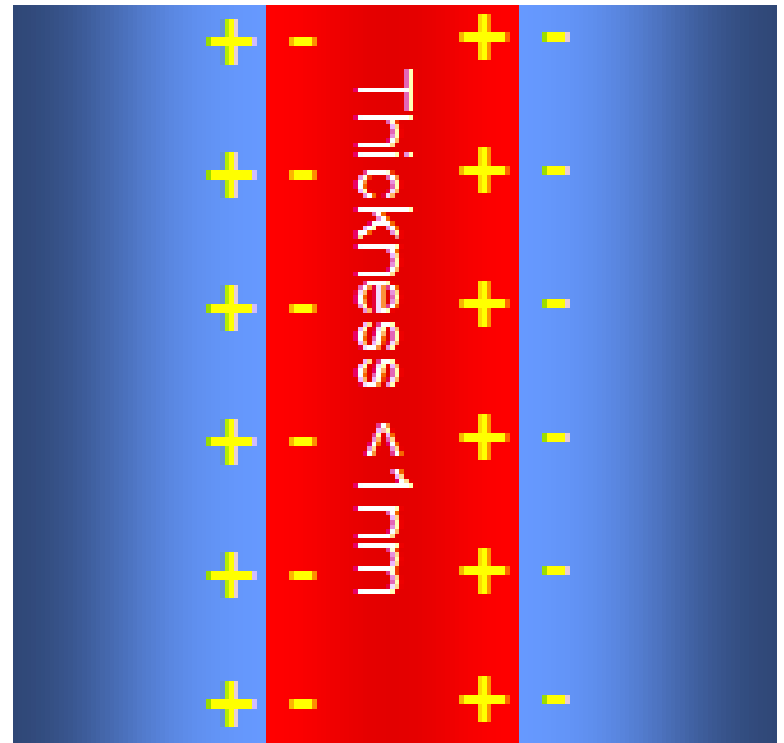
Capacitor



$$C \propto 1/\text{thickness}$$

$$E = \frac{1}{2} CV^2$$

Supercapacitor



Electrolyte solution

# Energy Storage Devices



## Batteries



Electrolyte solution

Capacitors and Super-capacitors are surface Storage

Batteries are bulk Storage



# Materials Challenges



Reactions occur at the electrode surfaces

- ✓ We want to get as high a surface area as possible
  - Need to have ions and electrons together for reactions to occur
  - However
    - e.g. Nanoparticles behave differently than bulk materials
    - Energy of the reactions also depend on the surface properties
- ✓ Electrons must still be able to get outside the cell
  1. Electron resistance cannot be too high
  2. Separator must be robust and allow rapid transfer of ions
  3. Fundamental materials properties need to be understood





# Materials Requirements

- Large reversible capacity
- Small irreversible capacity
- Desirable charge profile
- Desirable kinetics (rate capability)
- Long cycle and calendar life
- Ease of processing
- Safety
- Compatibility with electrolyte and binder systems
- Low cost



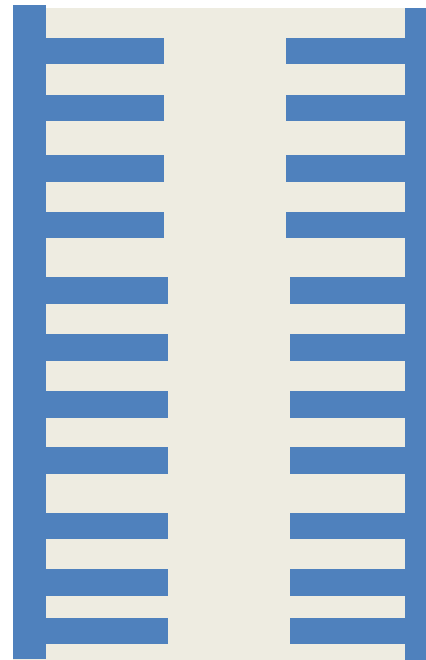
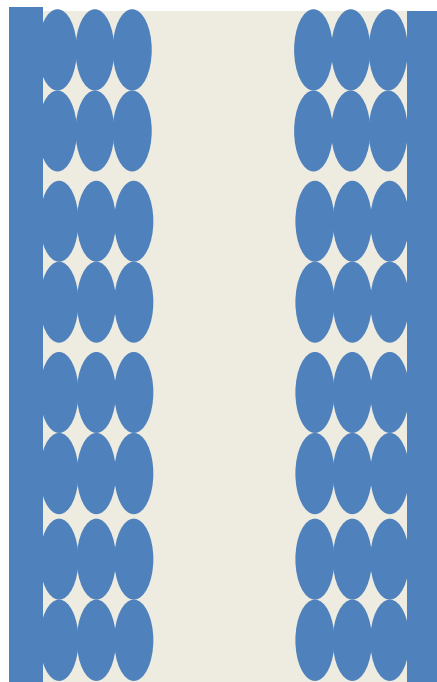
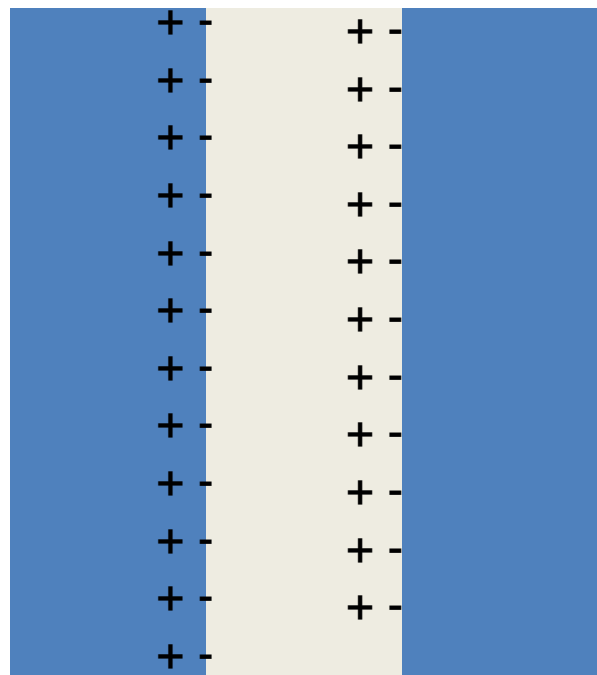
# Nanotechnology For Energy Storage



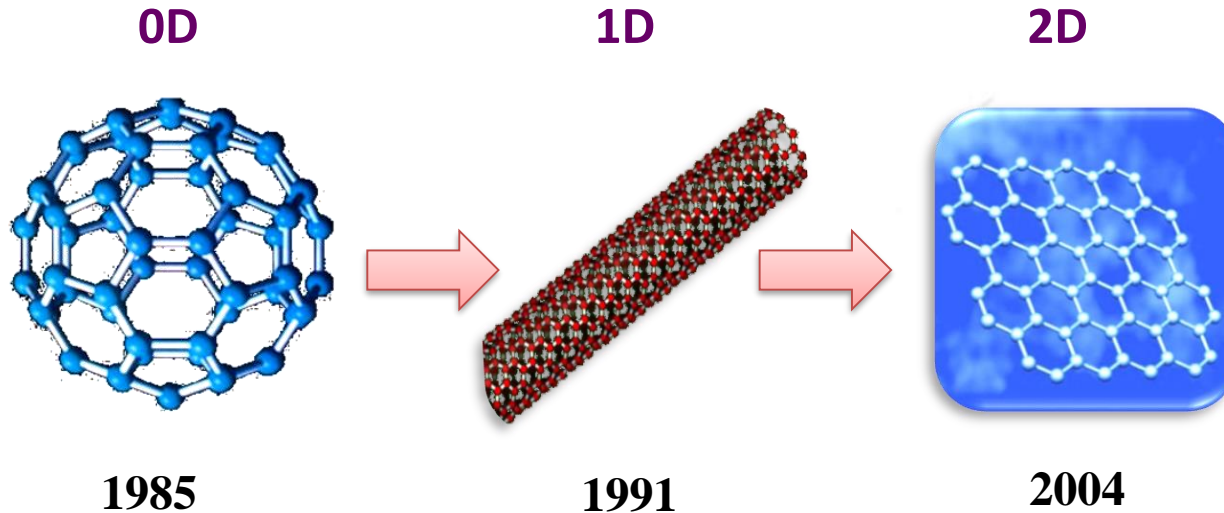
Bulk

Nanoparticles

Nanowires

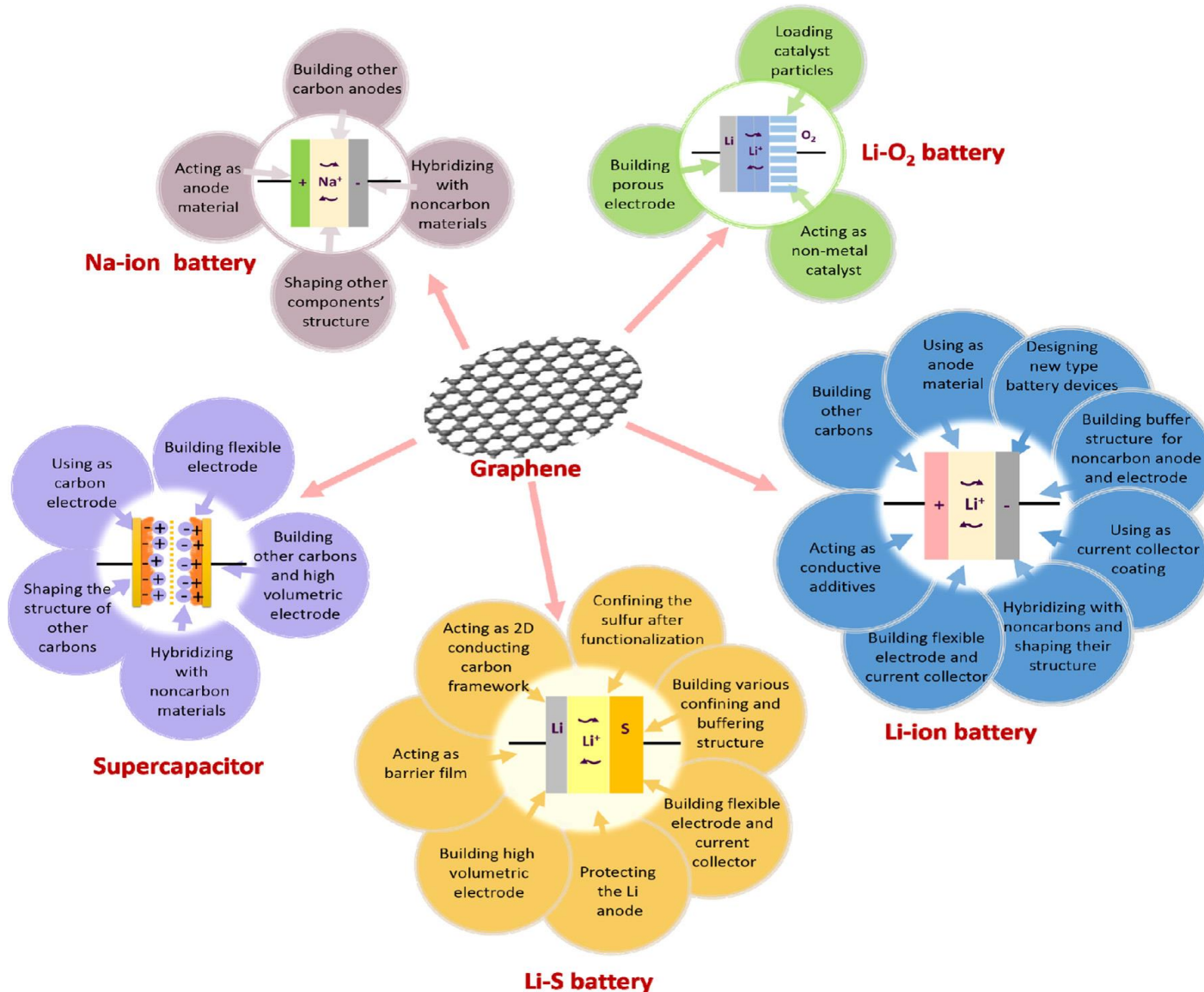


# Background on graphene

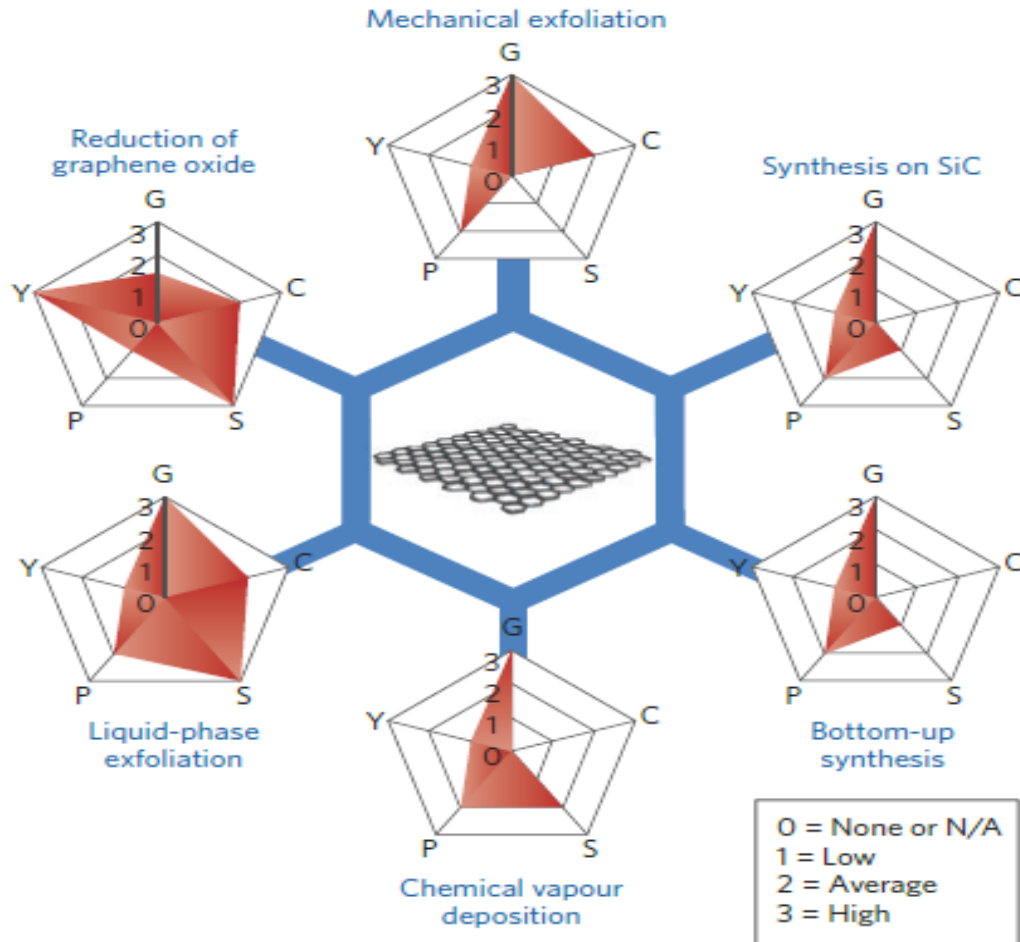


- **Electronic properties** (ballistic transport, high carrier mobility)
- **Mechanical properties** (~1.1 TPa modulus, ~130 Gpa fracture strength)
- **Thermal conductivity** (~5300 W m<sup>-1</sup> K<sup>-1</sup> in plane, highly anisotropic)
- **Optical transmittance** (97.7%)
- **Specific surface area** (theoretical: 2630 m<sup>2</sup> g<sup>-1</sup> )
- **Specific Capacitance** (theoretical: 550 Fg<sup>-1</sup> )

# Potential Applications

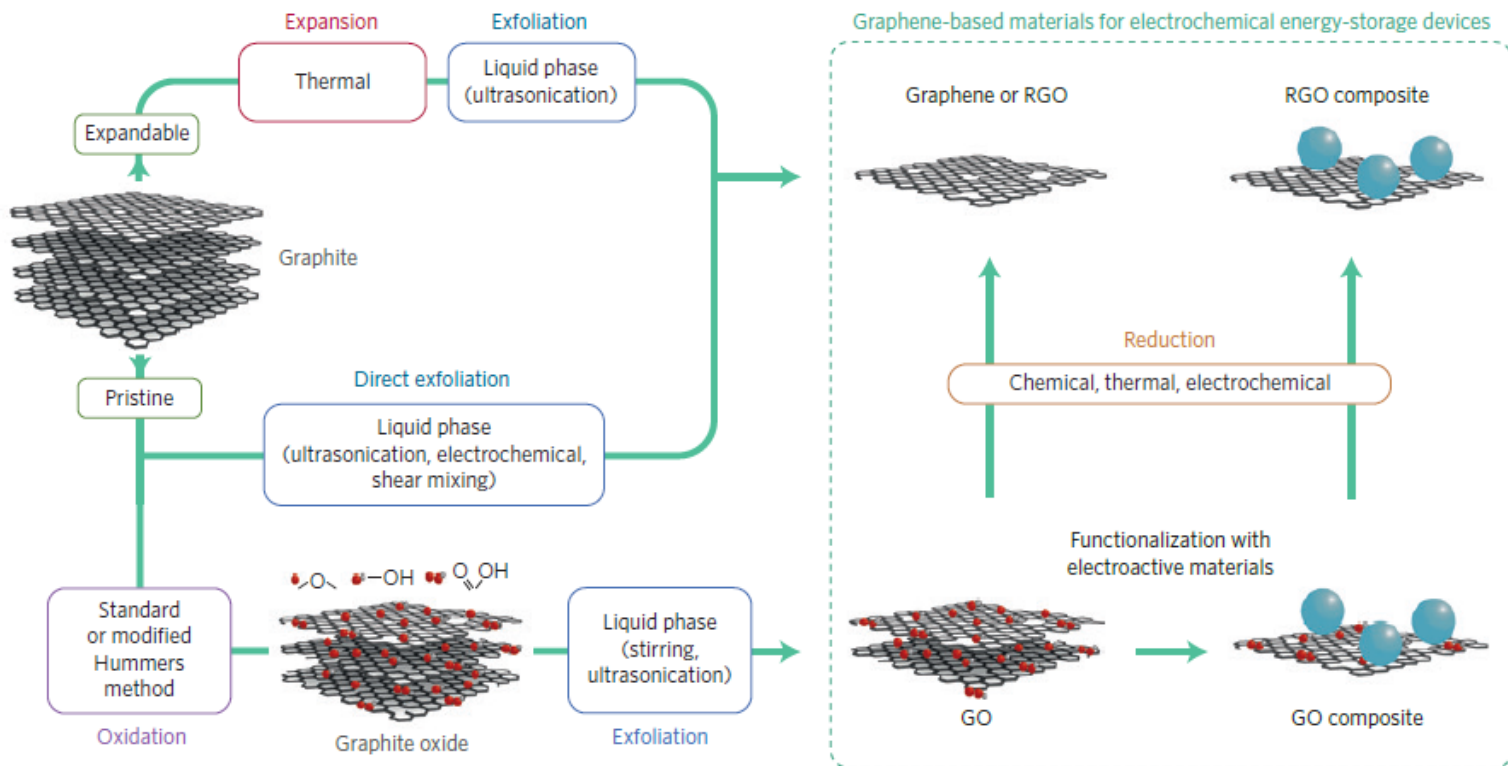


# Preparation of graphene



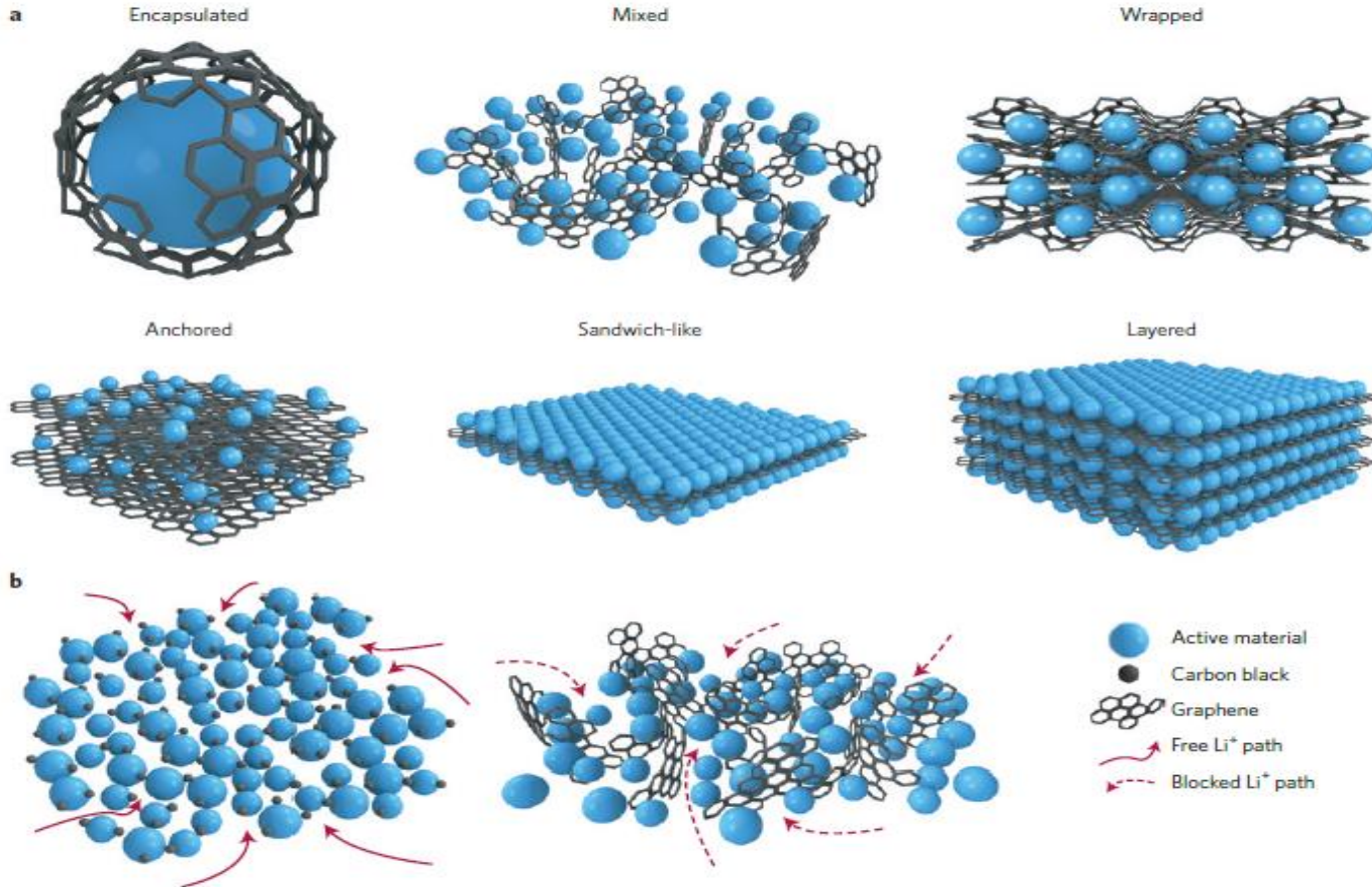
Schematic of the most common graphene production methods. Each method has been evaluated in terms of graphene quality (G), cost aspect (C; a low value corresponds to high cost of production), scalability (S), purity (P) and yield (Y) of the overall production process.

# Preparation of Graphene Based Composites



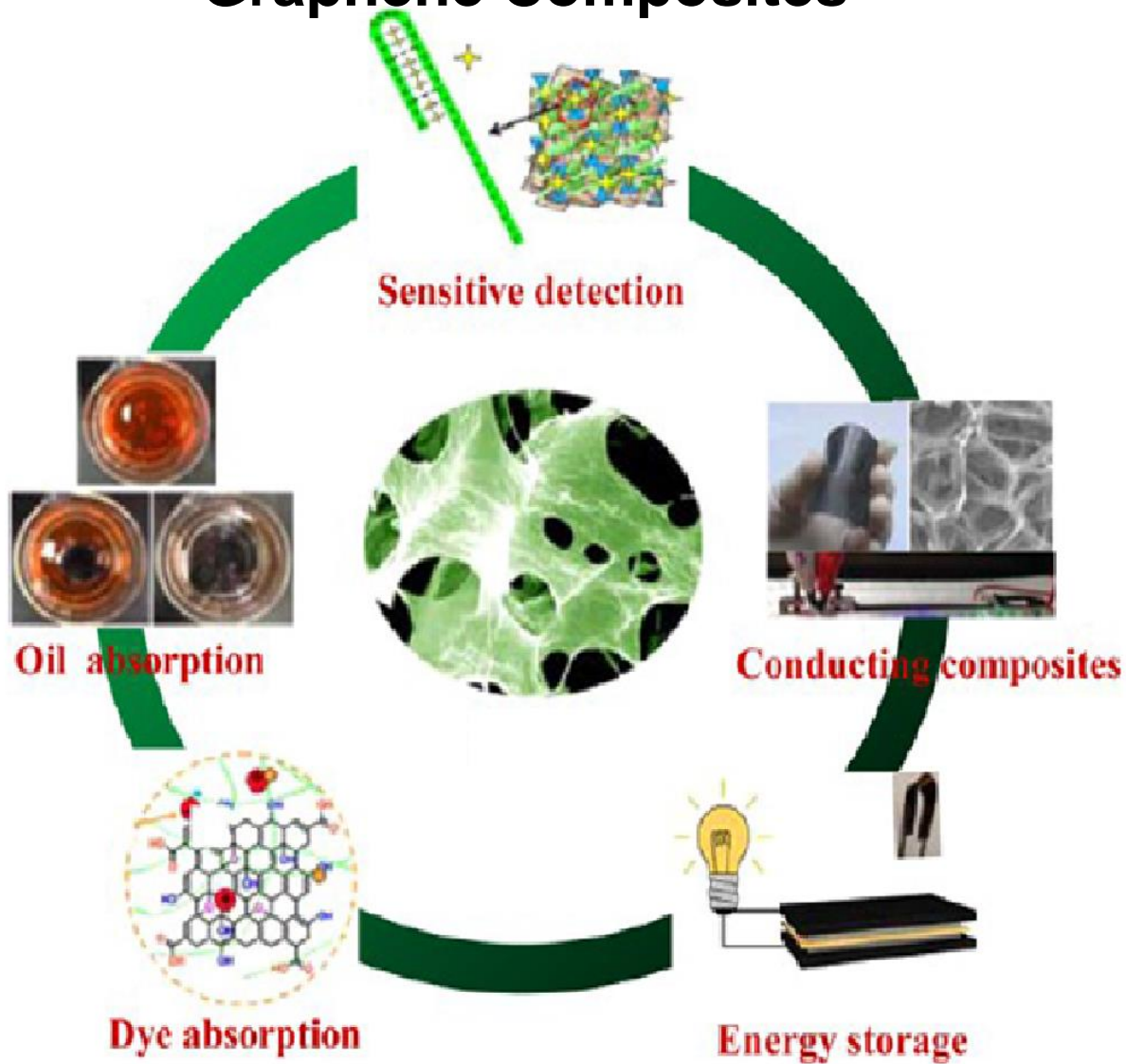
The most common synthetic pathways for producing graphene-based materials (GO, RGO, GO- and RGO-based compounds) for use as electrode active materials in EESDs.

# Preparation of Graphene Based Composites



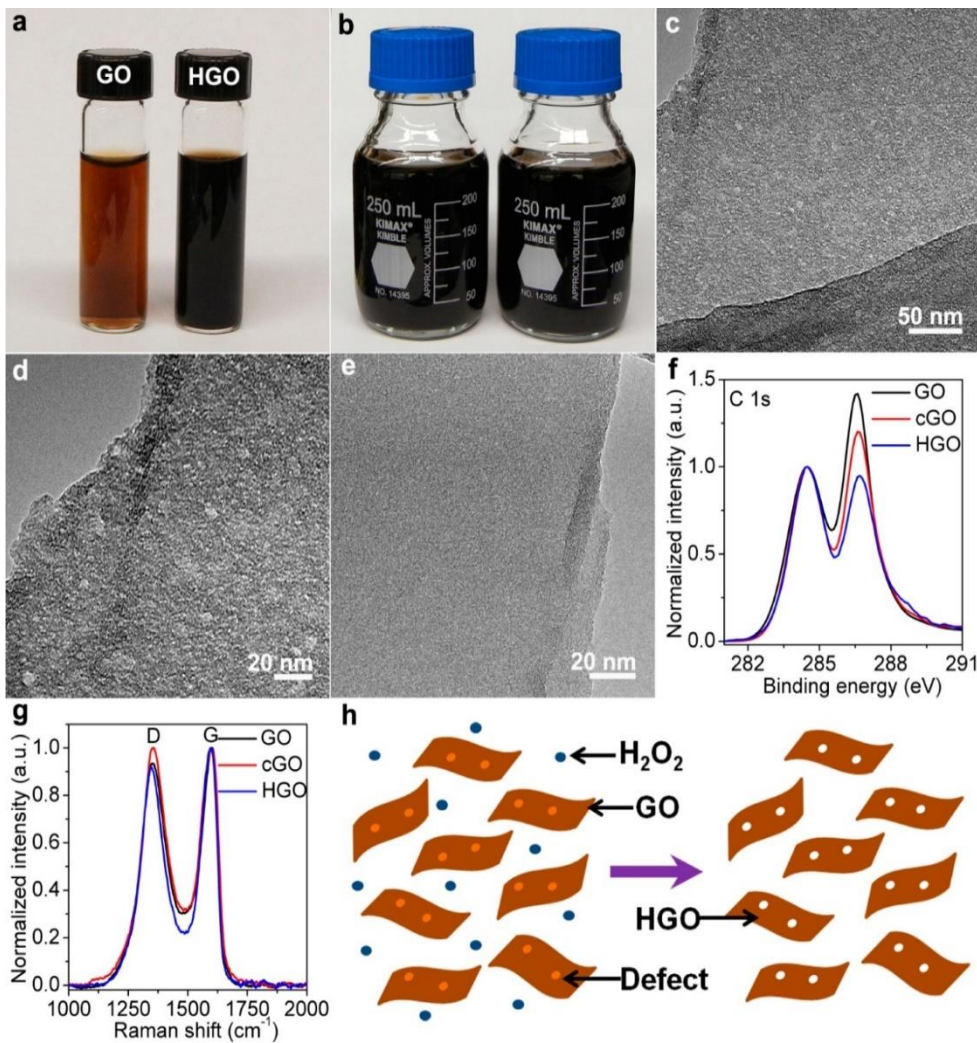
Structural models and a possible drawback of graphene composites.

# Potential Applications of 3D Graphene Composites





# Solution Processable Holey Graphene Oxide and Its Derived Macrostructures for High-Performance Supercapacitors

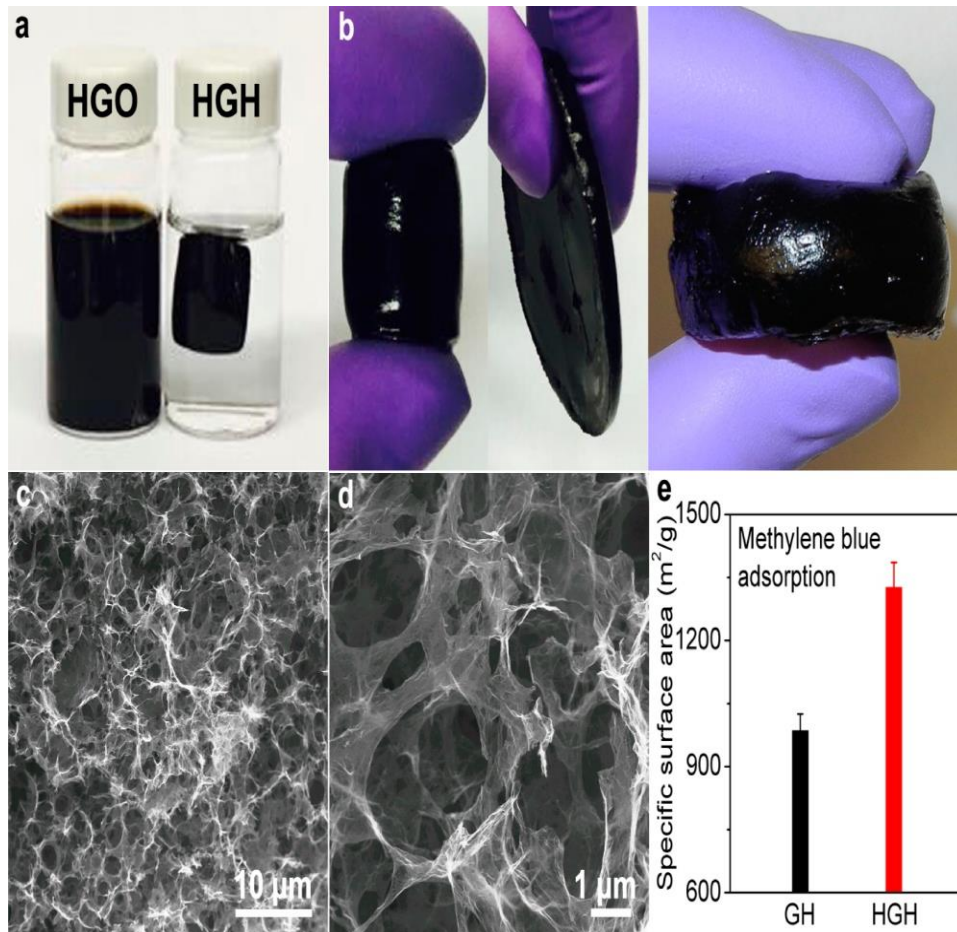


GO was prepared by oxidation of natural graphite powder according to the modified Hummers' method.

HGO was prepared according to the following procedure: Typically, 5 mL 30% H<sub>2</sub>O<sub>2</sub> aqueous solution was mixed with 50 mL 2 mg/mL GO aqueous dispersion and then heated at 100 °C for 4 h under stirring.

HGHs were prepared according to the following procedure: 0.5 mL 1 M sodium ascorbate aqueous solution was added into 10 mL 2 mg/mL HGO aqueous dispersion and then the homogeneous mixture was heated at 100 °C for 2 h without stirring

# Solution Processable Holey Graphene Oxide and Its Derived Macrostructures for High-Performance Supercapacitors

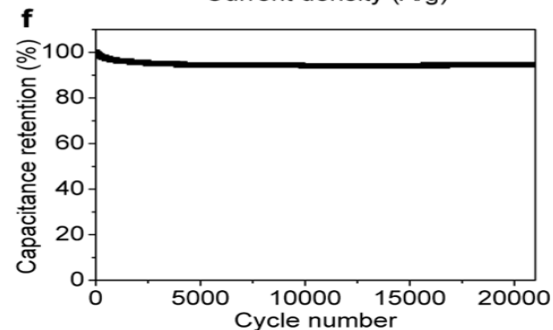
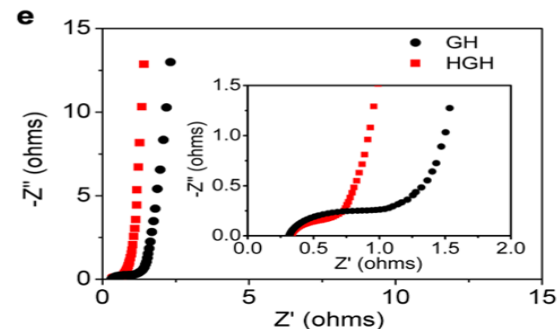
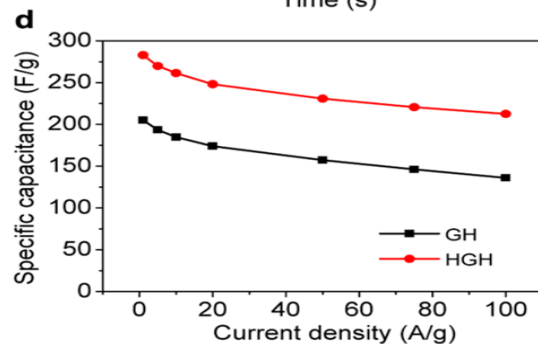
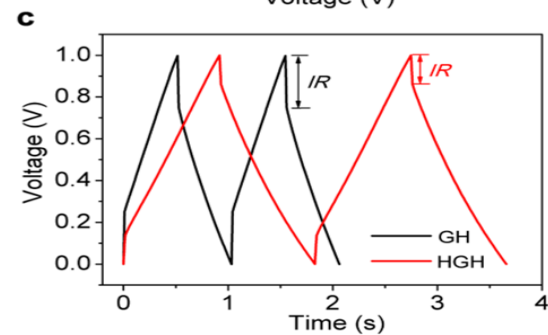
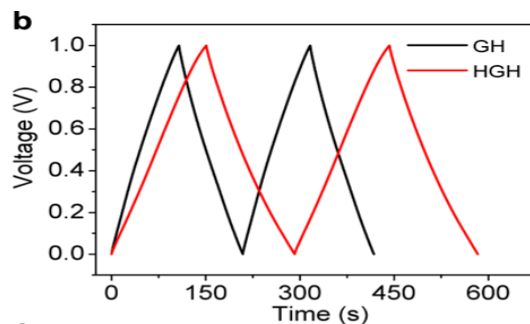
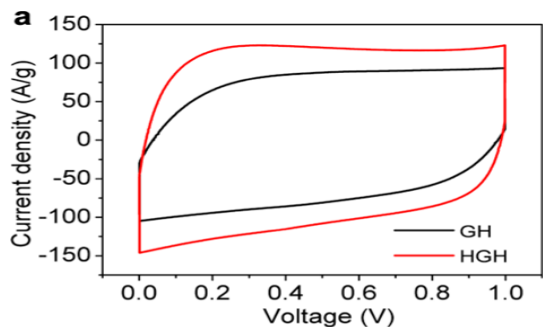


With the flexible processability of HGO dispersion, **the sizes and shapes of HGH** can be easily tailored by changing the type of reactors. The freeze-dried HGH showed an **interconnected 3D porous network** with the pore size ranging from submicrometers to several micrometers.

The HGH showed a very high accessible SSA of **~1330 m<sup>2</sup>/g**, ~34% higher than that of **GH (~990 m<sup>2</sup>/g)**,

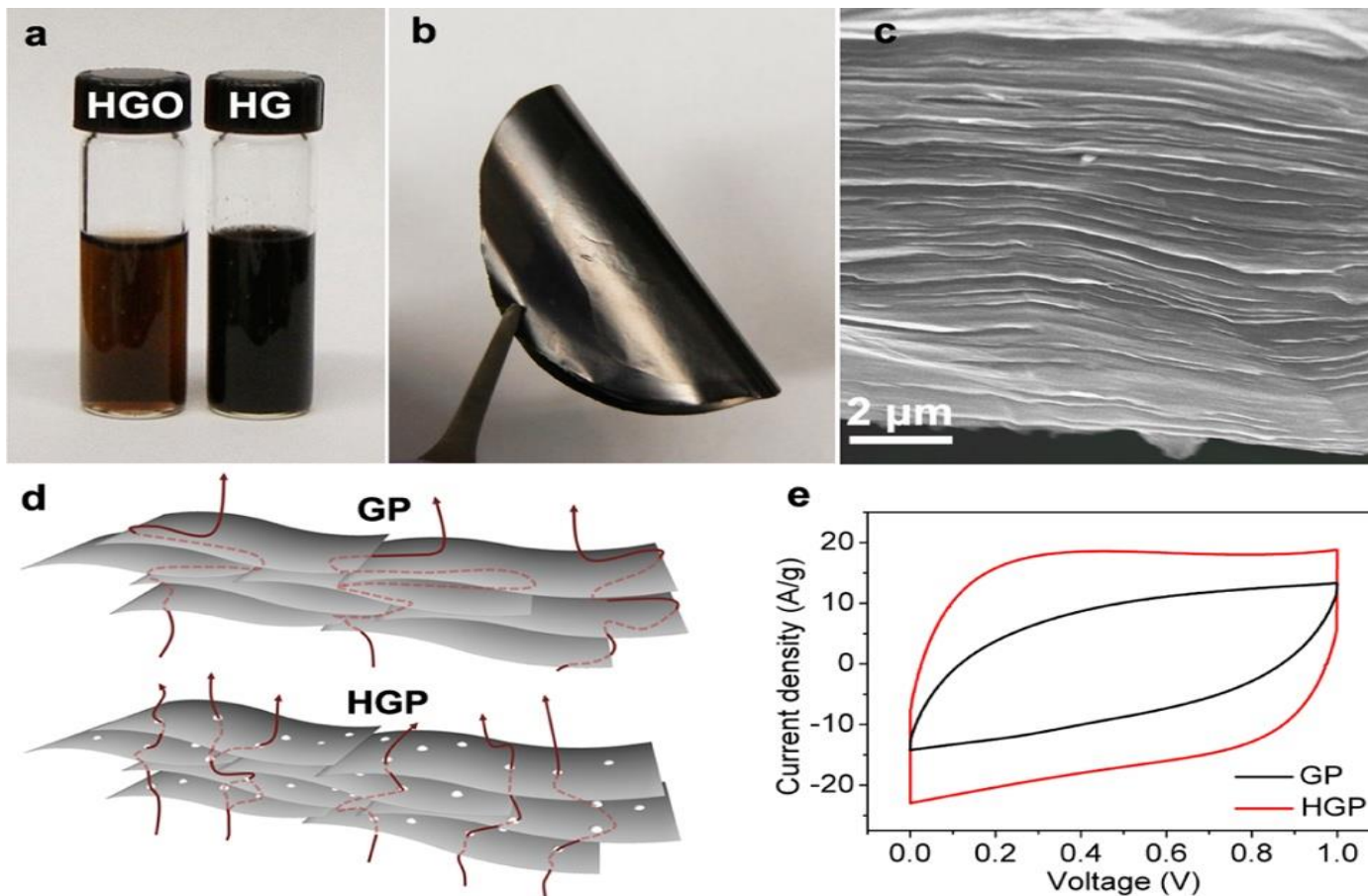
With the flexible processability of HGO dispersion, the sizes and shapes of HGH can be easily tailored by changing the type of reactors.

# Solution Processable Holey Graphene Oxide and Its Derived Macrostructures for High-Performance Supercapacitors



The HGH exhibited a specific capacitance of **283 F/g** at a current density of 1 A/g, 38% higher than that of GH (205 F/g)

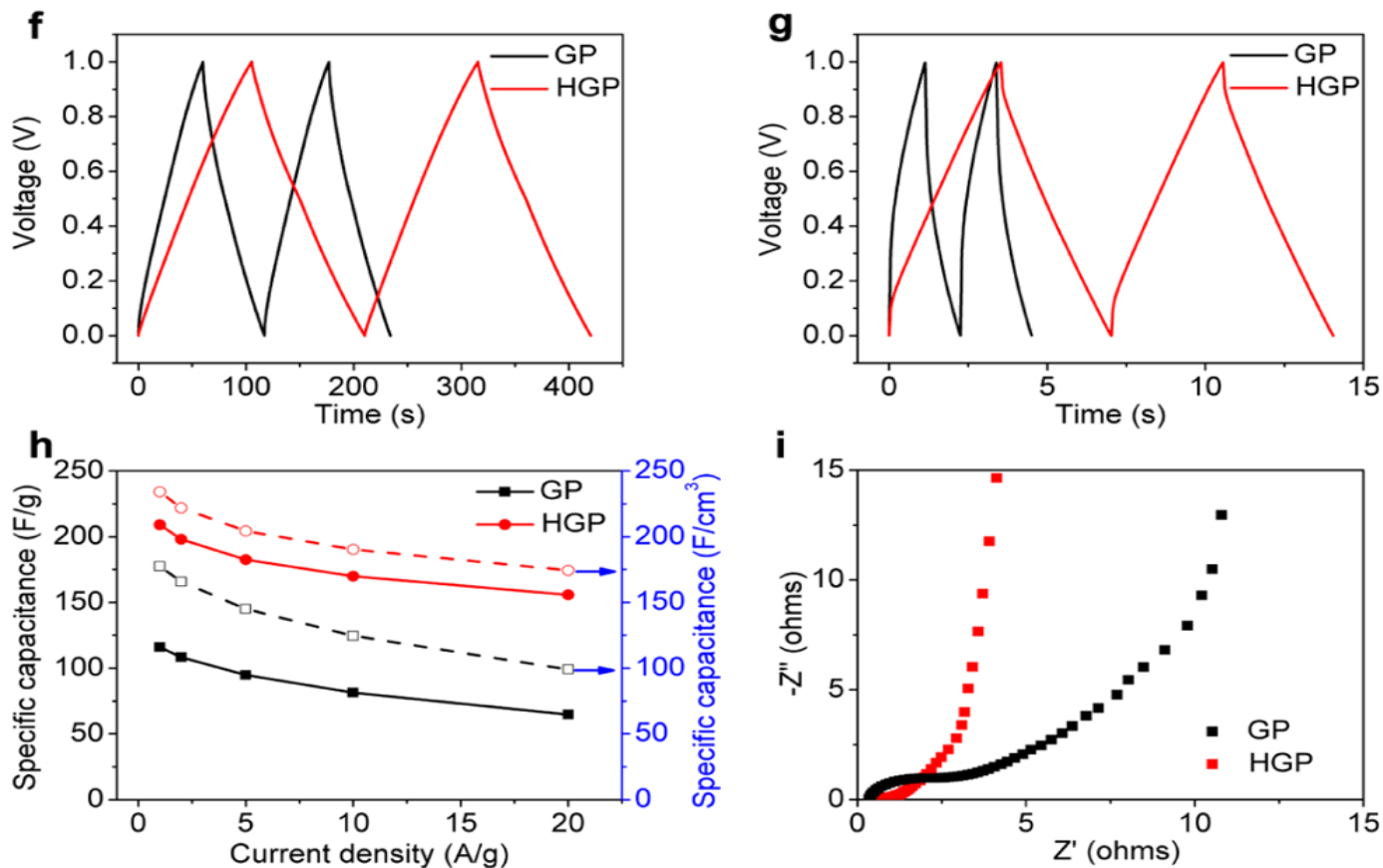
# Solution Processable Holey Graphene Oxide and Its Derived Macrostructures for High-Performance Supercapacitors



By chemical reduction of the HGO dispersion, we can also obtain well-dispersed solution processable HG which can be further assembled into large-area flexible HGP with a compact layered structure via a flow-directed self assembly strategy .

*Nano Lett.* 2015, 15, 4605.

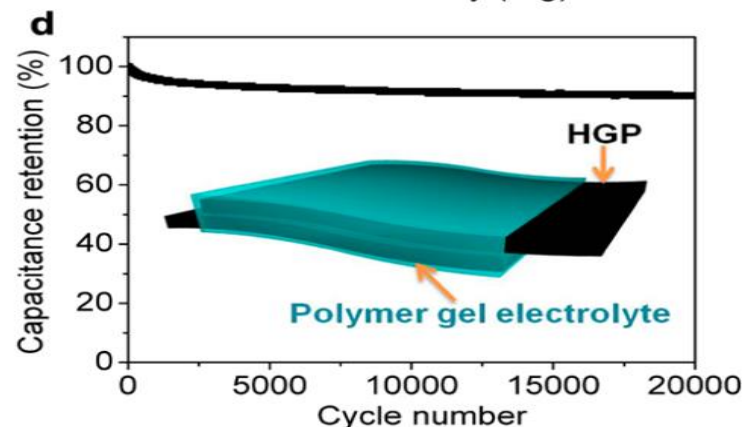
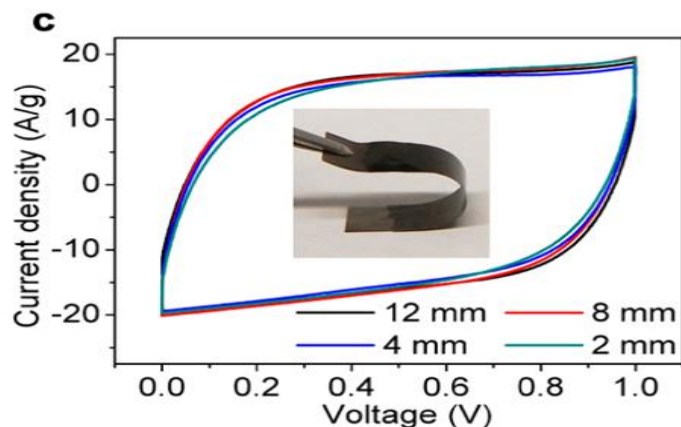
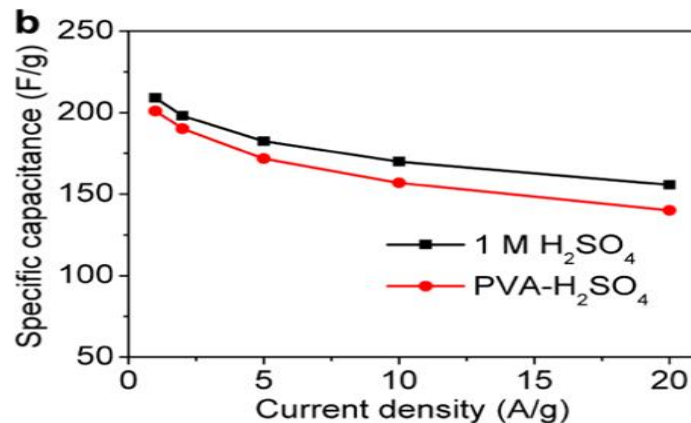
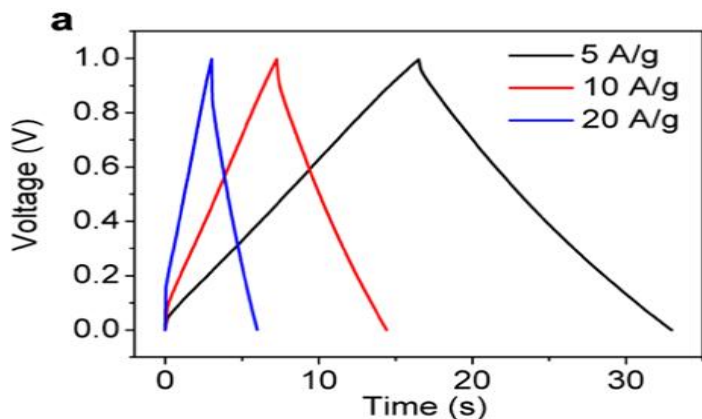
# Solution Processable Holey Graphene Oxide and Its Derived Macrostructures for High-Performance Supercapacitors



The HGP exhibited a specific capacitance of **209 F/g** at 1 A/g, respectively, considerably higher than those of GP (116 F/g). With a high packing density of 1.12 g/cm<sup>3</sup>, the HGP could deliver an ultrahigh volumetric capacitance of **234 F/cm<sup>3</sup>**.

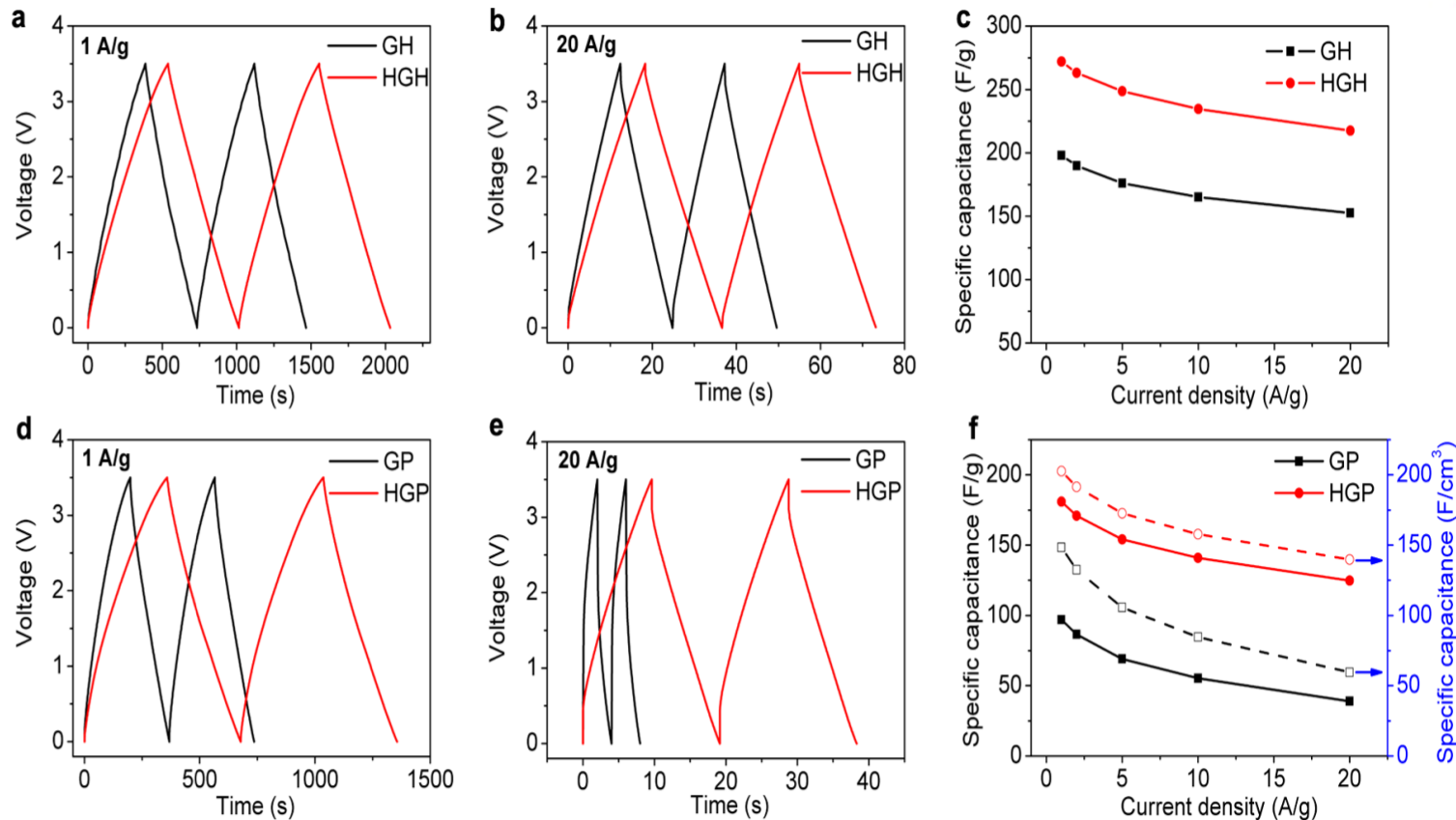
Nano Lett. 2015, 15, 4605.

# Solution Processable Holey Graphene Oxide and Its Derived Macrostructures for High-Performance Supercapacitors



The entire flexible solid-state supercapacitor based on HGP showed a superior volumetric capacitance of **34 F/cm<sup>3</sup>**, which outperform most of previous reported results.

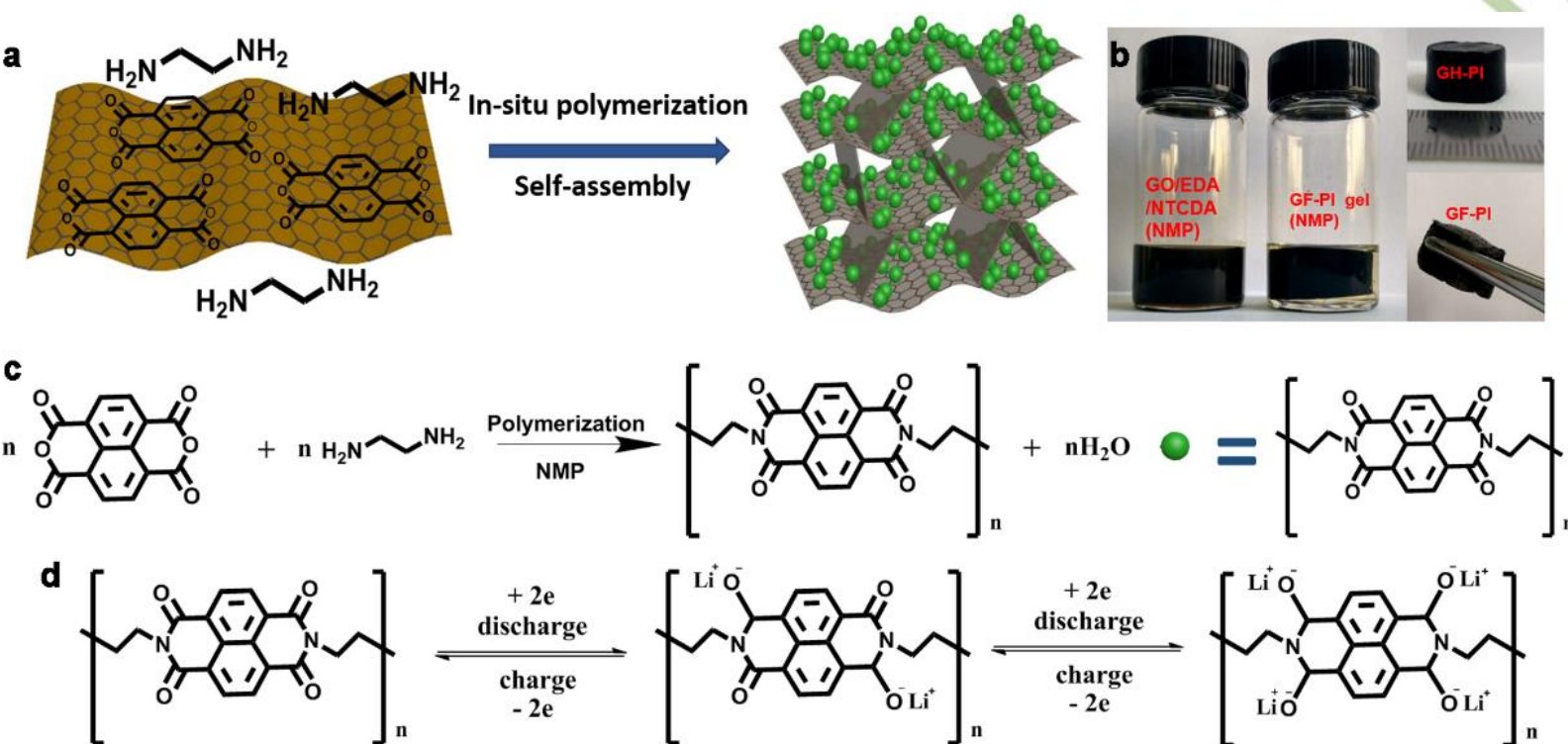
# Solution Processable Holey Graphene Oxide and Its Derived Macrostructures for High-Performance Supercapacitors



With EMIMBF<sub>4</sub>/AN as organic electrolyte, a high energy density of **116 and 77 Wh/kg** can be achieved for HGH and HGP, respectively. Moreover, the HGP could show a high volumetric capacitance of **203 F/cm<sup>3</sup>** and a high volumetric energy density of **86 Wh/L**.



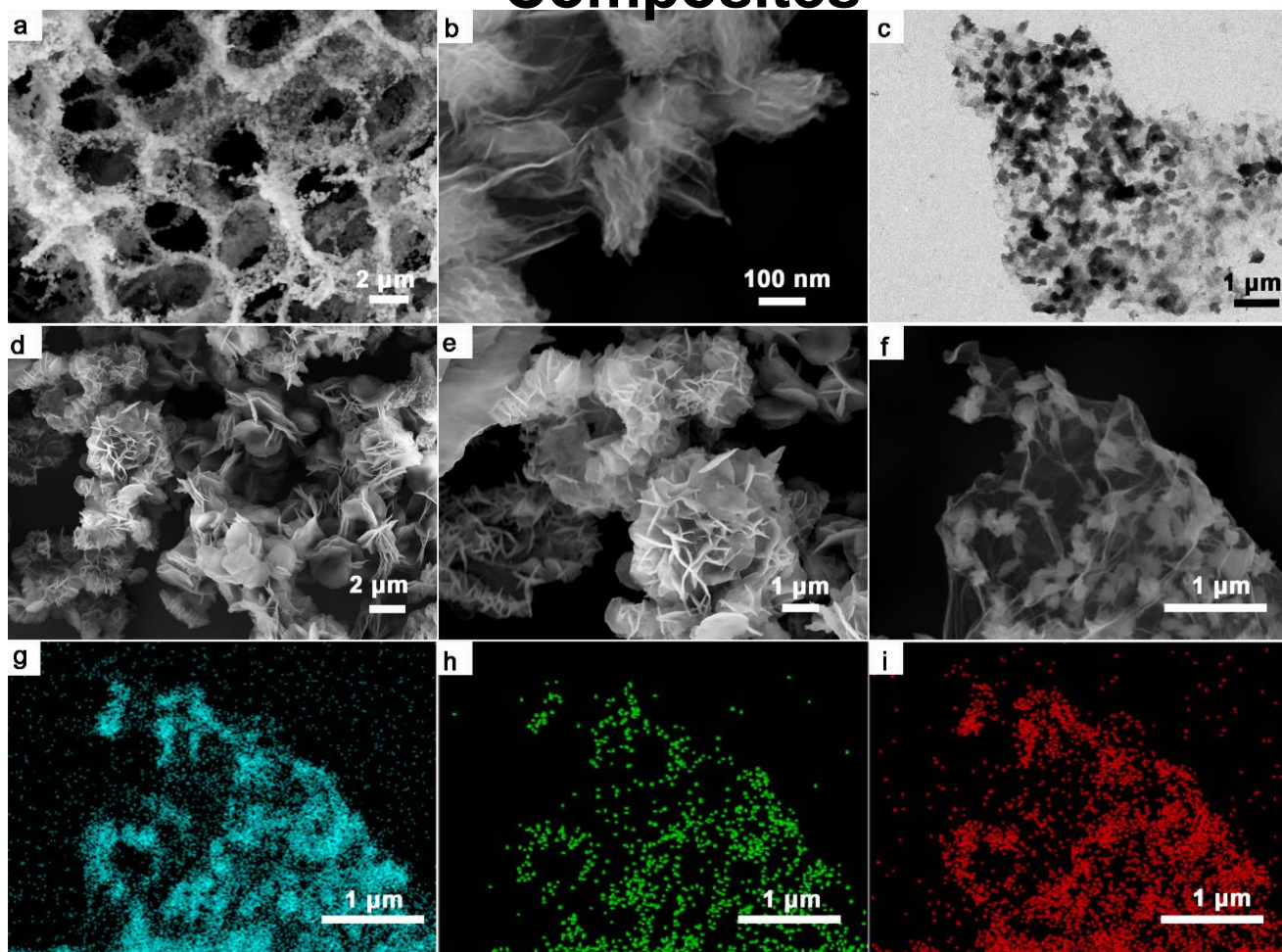
# Three-Dimensional Graphene/Polyimide Composites



Schematic of the preparation process of GF-PI; (b) Photographs of precursor solution of GO, naphthalenetetracarboxylic dianhydride (NTCDA) and ethanediamine (EDA), the resultant GF-PI gel in N-Methyl-2-pyrrolidone (NMP) solution, the GH-PI with interior NMP exchanged by water, and the final GF-PI monolith after freeze-drying and annealing; (c) Synthetic route to PI; (d) Electrochemical redox mechanism of PI.

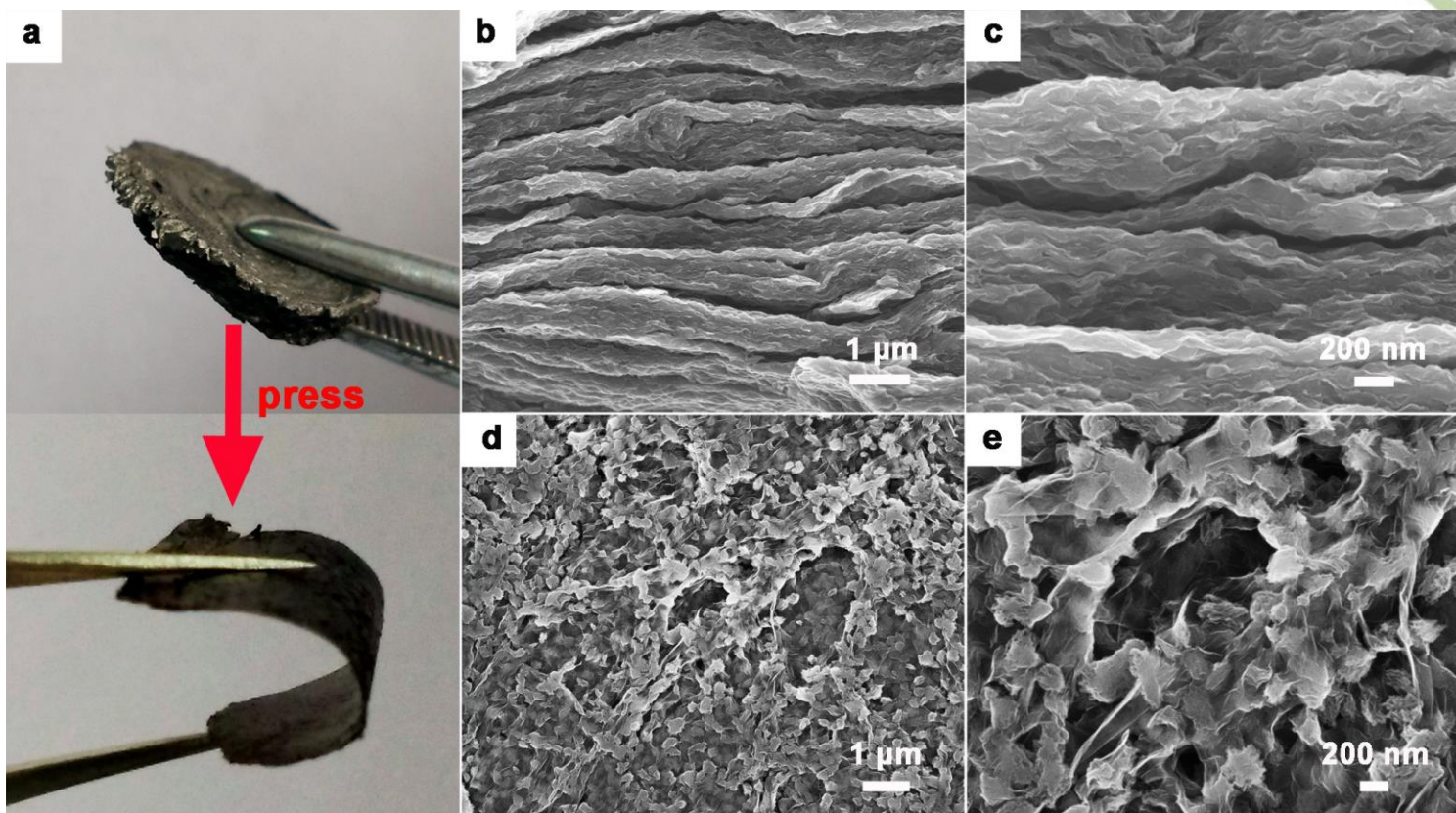


# Three-Dimensional Graphene/Polyimide Composites



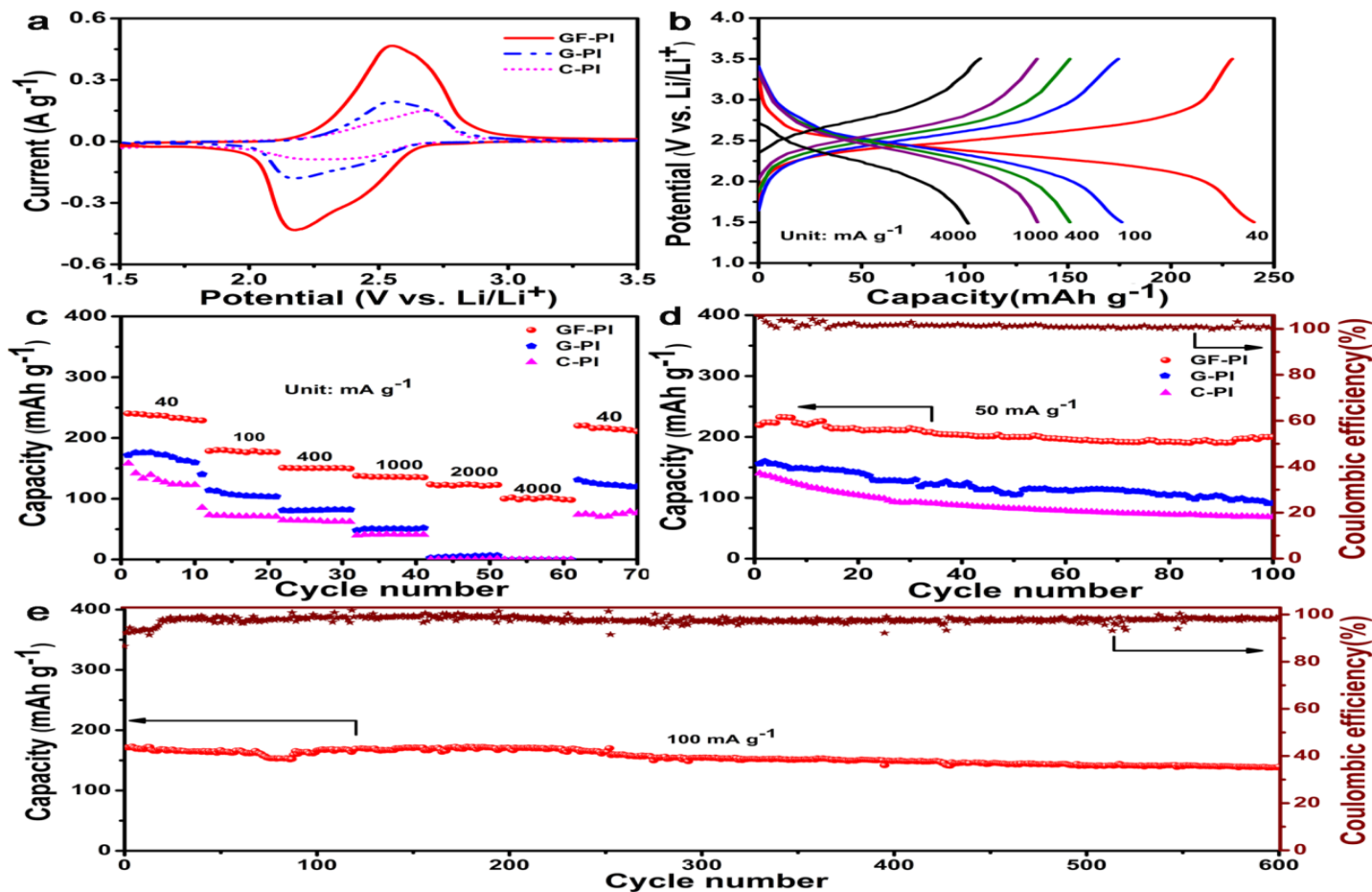
Low- and (b) high-magnification SEM images and (c) TEM image of GF-PI; (d) Low- and (e) high-magnification SEM images of pure PI; (f-i) Elemental mapping images of GF-PI (by SEM): Carbon (g), N (h), and O (i) distribution in the selected area.

# Three-Dimensional Graphene/Polyimide Composites



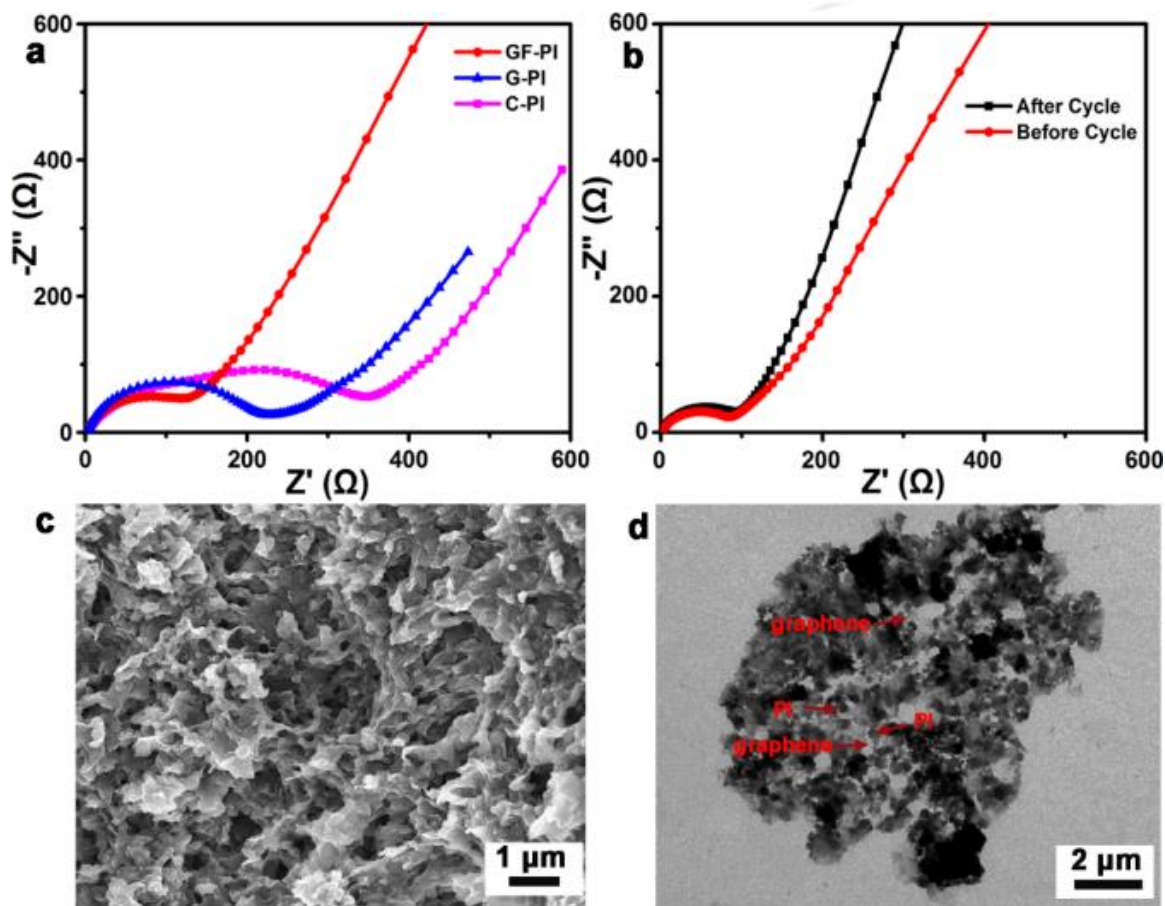
Preparation of a flexible binder-free GF-PI electrode. A piece of GF-PI (above) was mechanically pressed into a film electrode (down); SEM image of the side-view (b, c) and top-view (d, e) of the pressed GF-PI film.

# Three-Dimensional Graphene/Polyimide Composites



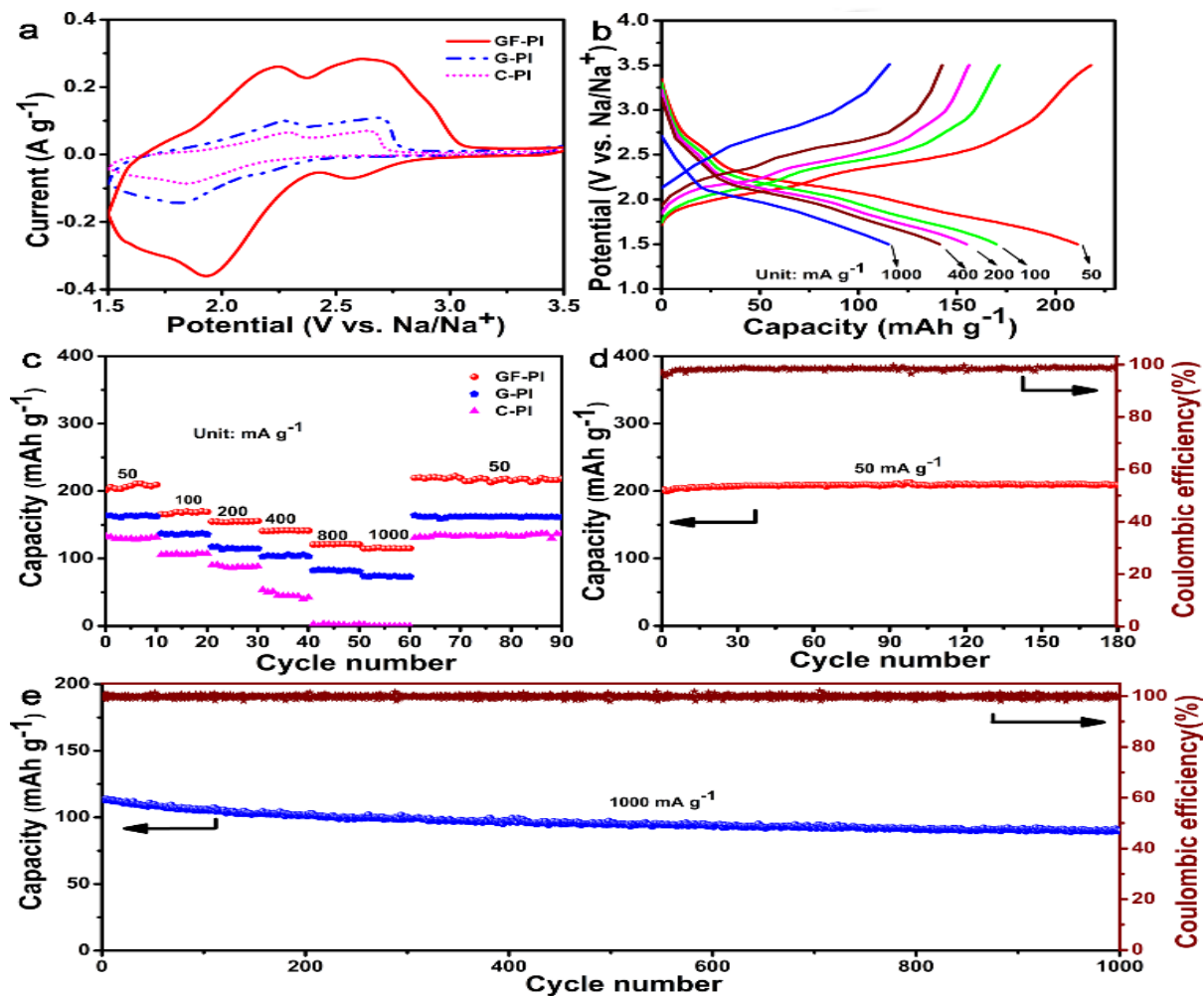
Electrochemical characterization of GF-PI as LIB cathode.

# Three-Dimensional Graphene/Polyimide Composites



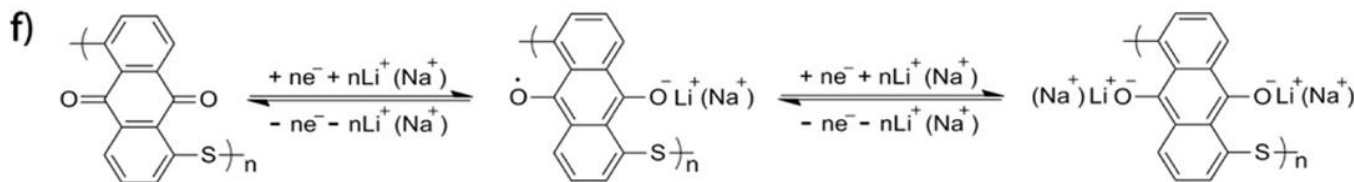
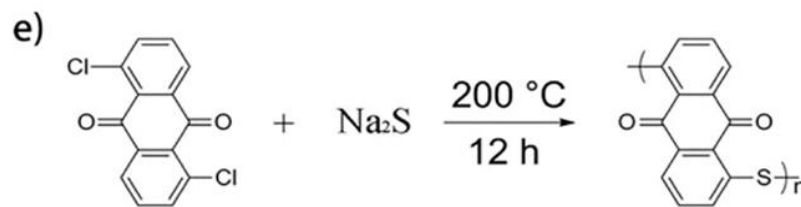
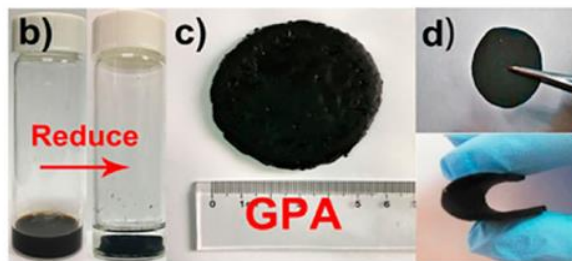
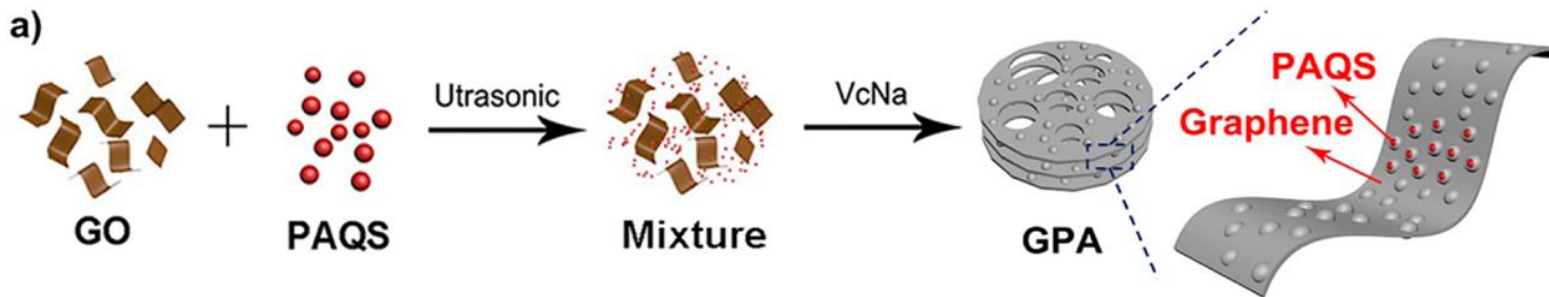
Nyquist plots for GF-PI, G-PI and C-PI electrodes; (b) Nyquist plots for GF-PI electrode before and after cycling test; (c) SEM and (d) TEM image of the GF-PI after 600 cycles.

# Three-Dimensional Graphene/Polyimide Composites



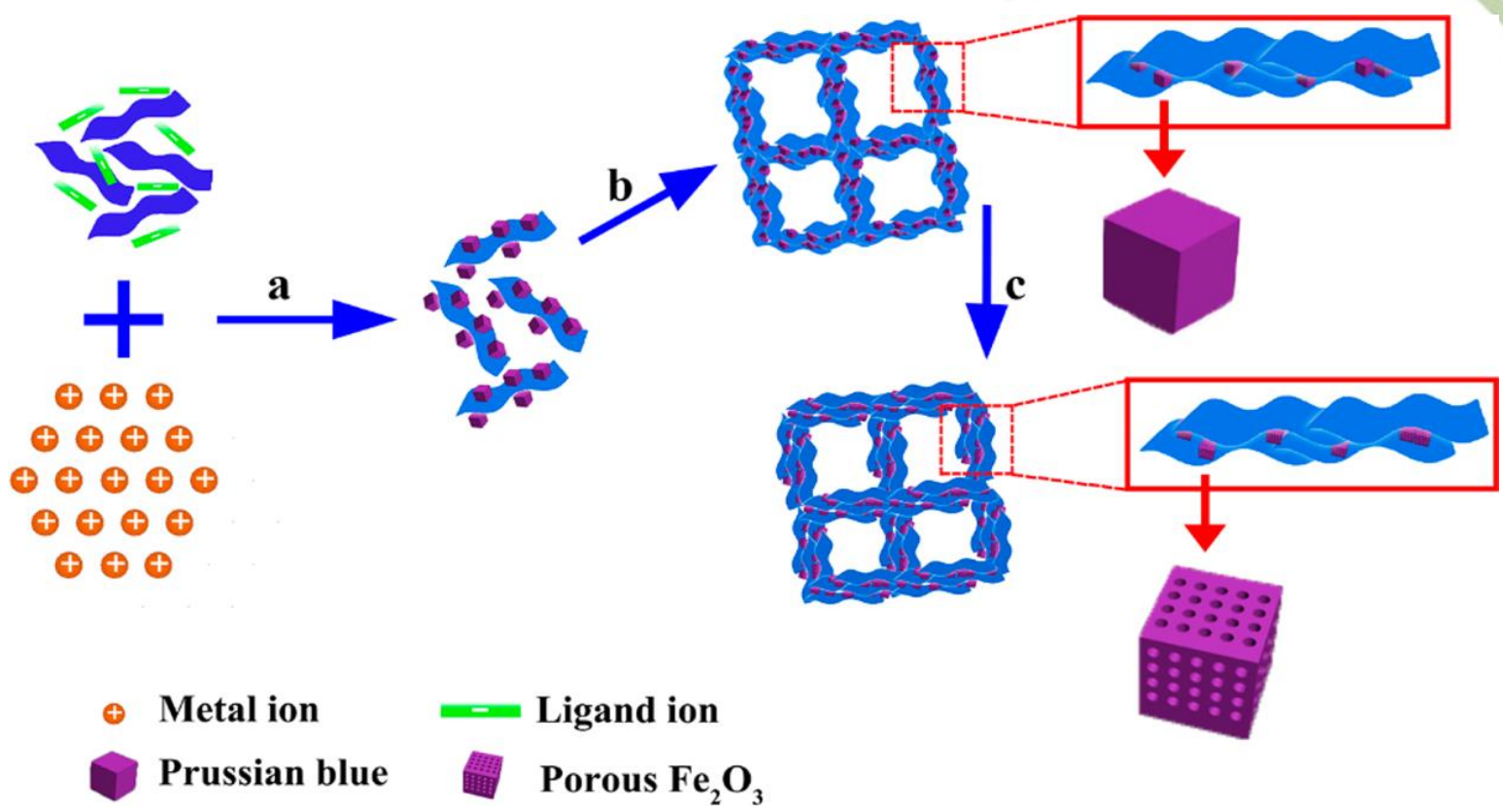
Electrochemical characterization of GF-PI as SIB cathode.

# Three-Dimensional Graphene/Poly (anthraquinonyl sulfide) (PAQS) Composites



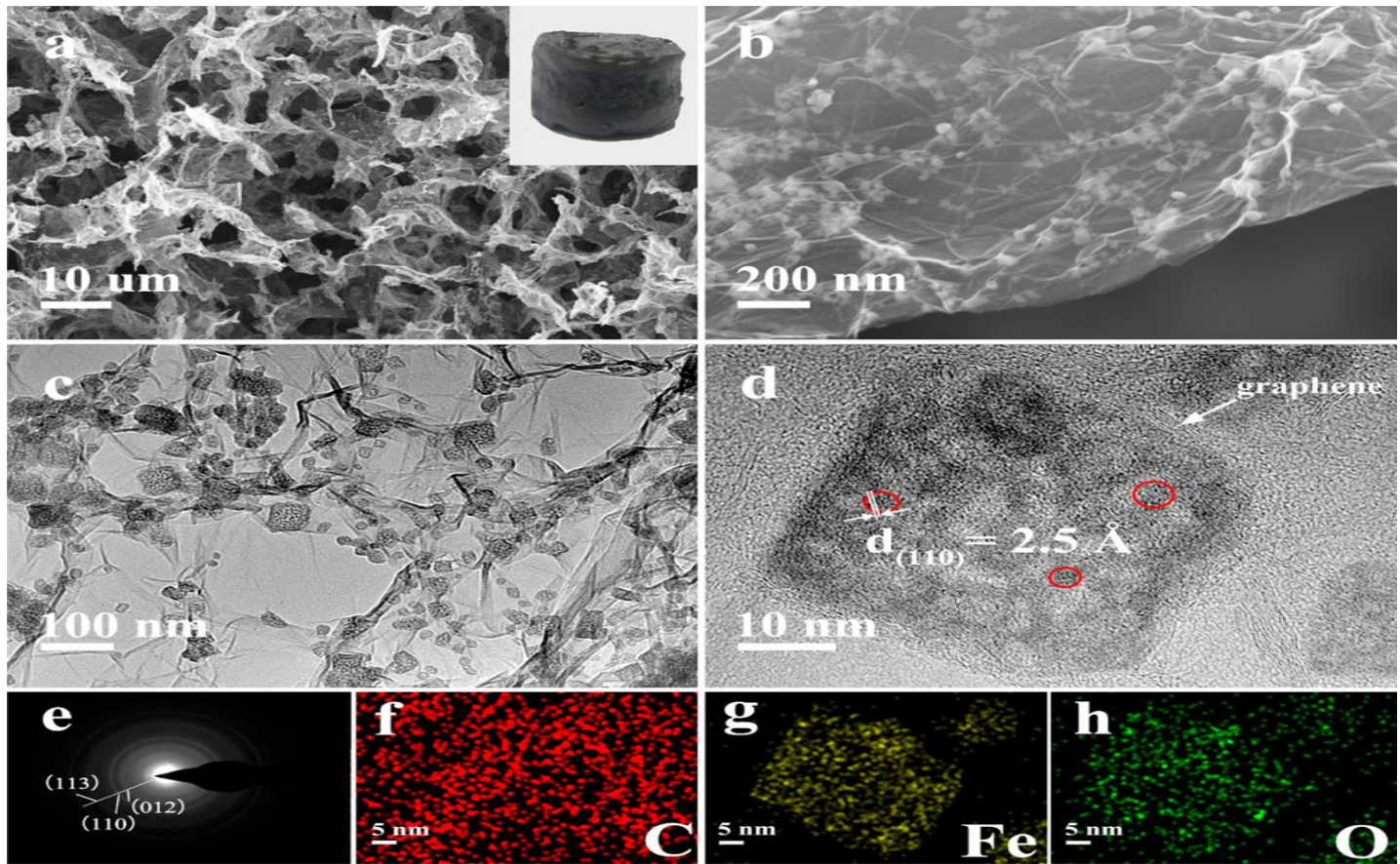
Schematic process of GPA. (b) Homogenous mixture of GO and poly (anthraquinonyl sulfide) PAQS in NMP and reduction and self-assembly of GO to form 3D graphene framework with encapsulated PAQS particles. (c) Freeze-dried aerogel of graphene/PAQS composite. (d) Preparation of a flexible binder-free GPA film electrode by mechanical pressing. (e) Synthetic route to PAQS. (f) Reversible redox mechanisms of PAQS in LIBs and SIBs.

# Porous $\text{Fe}_2\text{O}_3$ Nanoframeworks Encapsulated within Three-Dimensional Graphene



Schematic illustration of the preparation procedure of 3DG/ $\text{Fe}_2\text{O}_3$ . (a) Excessive metal ions induced combination, (b) self-assembly and spatially confined Ostwald ripening, and (c) thermal treatment under air condition.

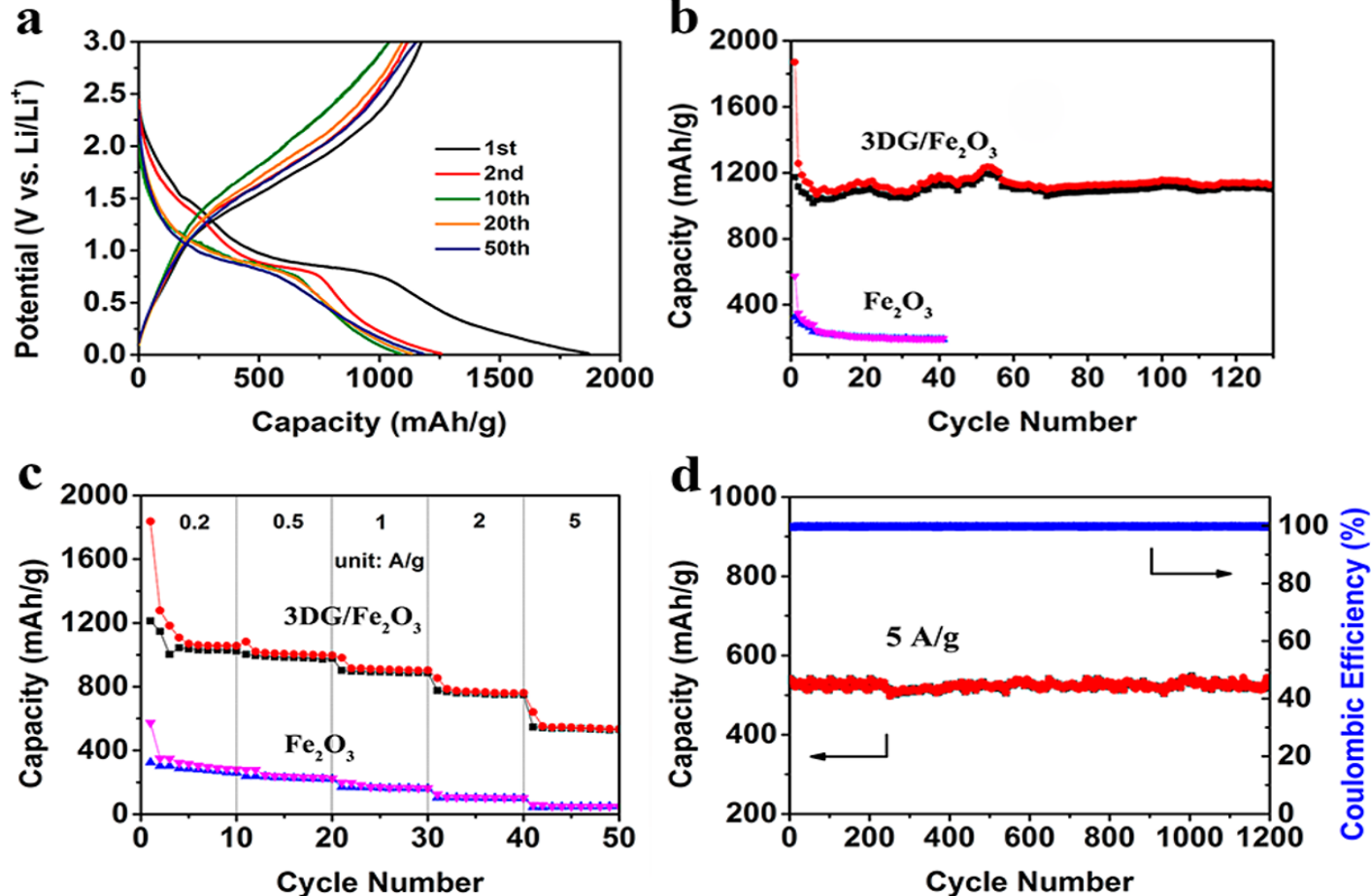
# Porous $\text{Fe}_2\text{O}_3$ Nanoframeworks Encapsulated within Three-Dimensional Graphene



SEM and (c) TEM images of the microstructure of 3DG/ $\text{Fe}_2\text{O}_3$  prepared by annealing of 3DG/PB at 250  $^\circ\text{C}$  for 2h. (d) HR-TEM image of 3DG/ $\text{Fe}_2\text{O}_3$  and (e) the SAED pattern showing the poor crystallinity of  $\text{Fe}_2\text{O}_3$ . Elemental mapping images of C (f), Fe (g), and O (h) of 3DG/ $\text{Fe}_2\text{O}_3$  composite shown in (d).



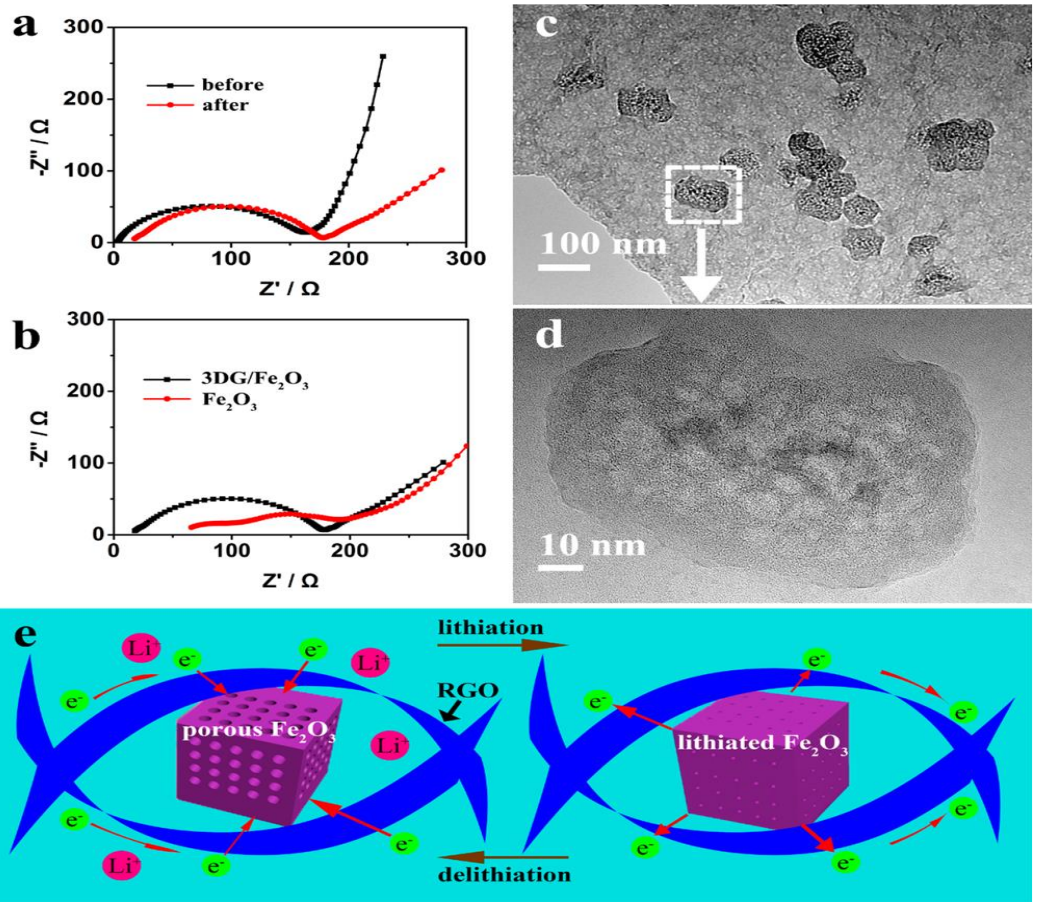
# Porous $\text{Fe}_2\text{O}_3$ Nanoframeworks Encapsulated within Three-Dimensional Graphene



Discharge/charge profiles of 3DG/Fe<sub>2</sub>O<sub>3</sub> at a current density of 0.2 A/g. The cycling performance at 0.2 A/g (b) and rate performance (c) of 3DG/Fe<sub>2</sub>O<sub>3</sub> and Fe<sub>2</sub>O<sub>3</sub>, respectively. (d) Cycling performance of 3DG/Fe<sub>2</sub>O<sub>3</sub> at 5 A/g.

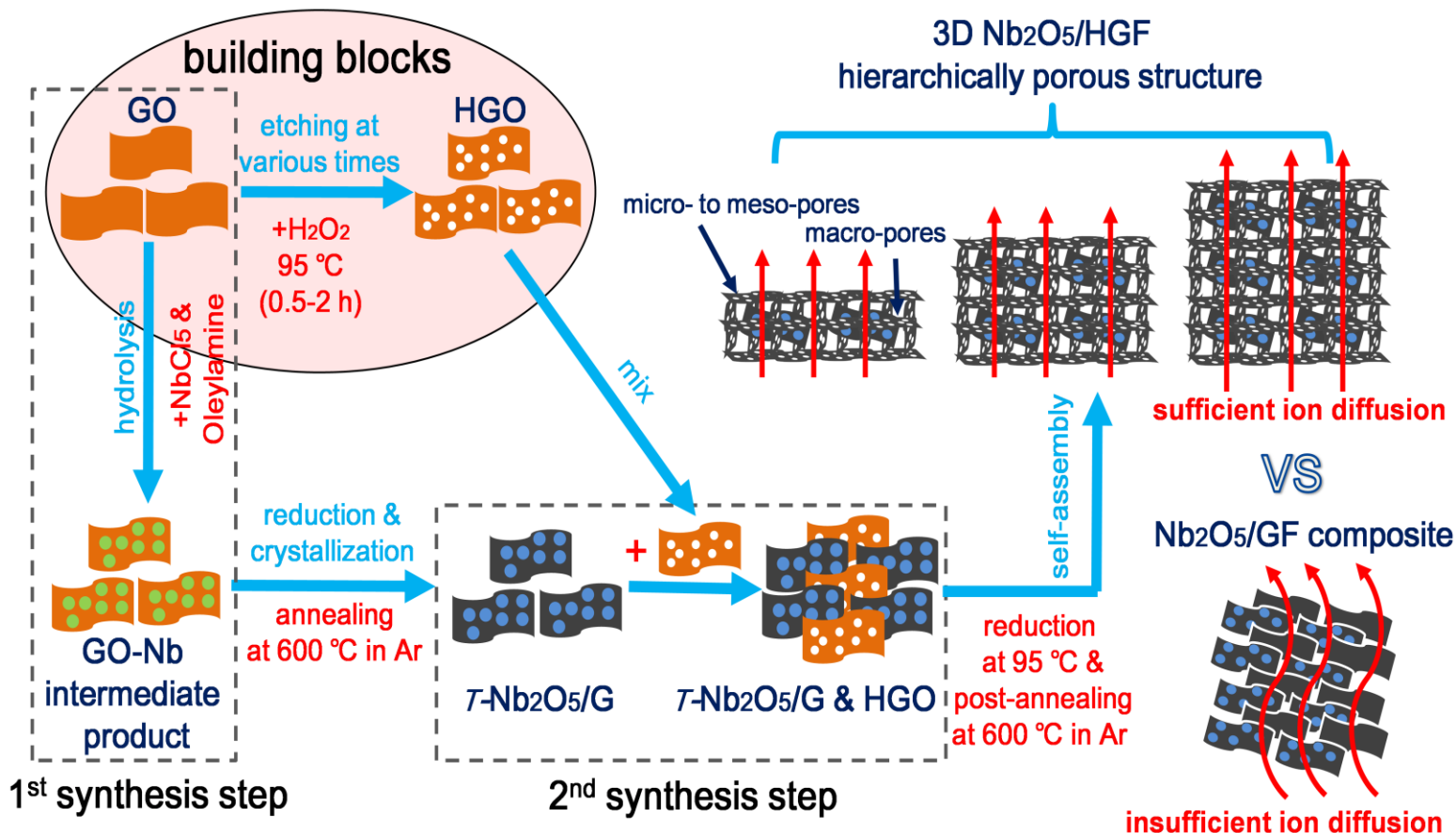


# Porous $\text{Fe}_2\text{O}_3$ Nanoframeworks Encapsulated within Three-Dimensional Graphene

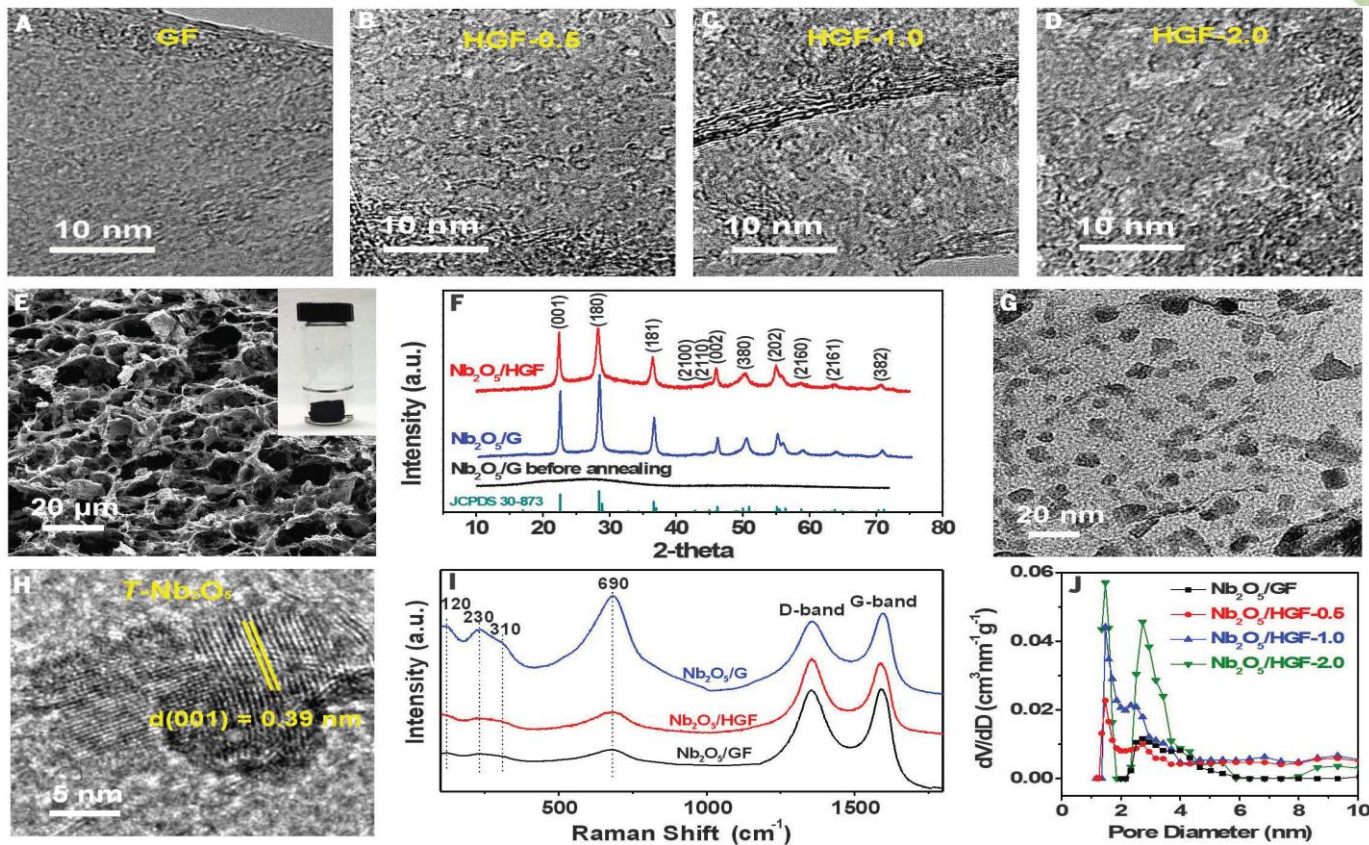


The Nyquist plots of 3DG/Fe<sub>2</sub>O<sub>3</sub> before and after cycling test. (b) The Nyquist plots of 3DG/Fe<sub>2</sub>O<sub>3</sub> and Fe<sub>2</sub>O<sub>3</sub> after cycling test. (c) TEM image of 3DG/Fe<sub>2</sub>O<sub>3</sub> and (d) higher magnification image of a single Fe<sub>2</sub>O<sub>3</sub> nanoframework under delithiation state after cycling test. (e) The schematic of delithiation/lithiation reactions of 3DG/Fe<sub>2</sub>O<sub>3</sub>

# Three-Dimensional Holey-Graphene/Niobia Composite Architectures

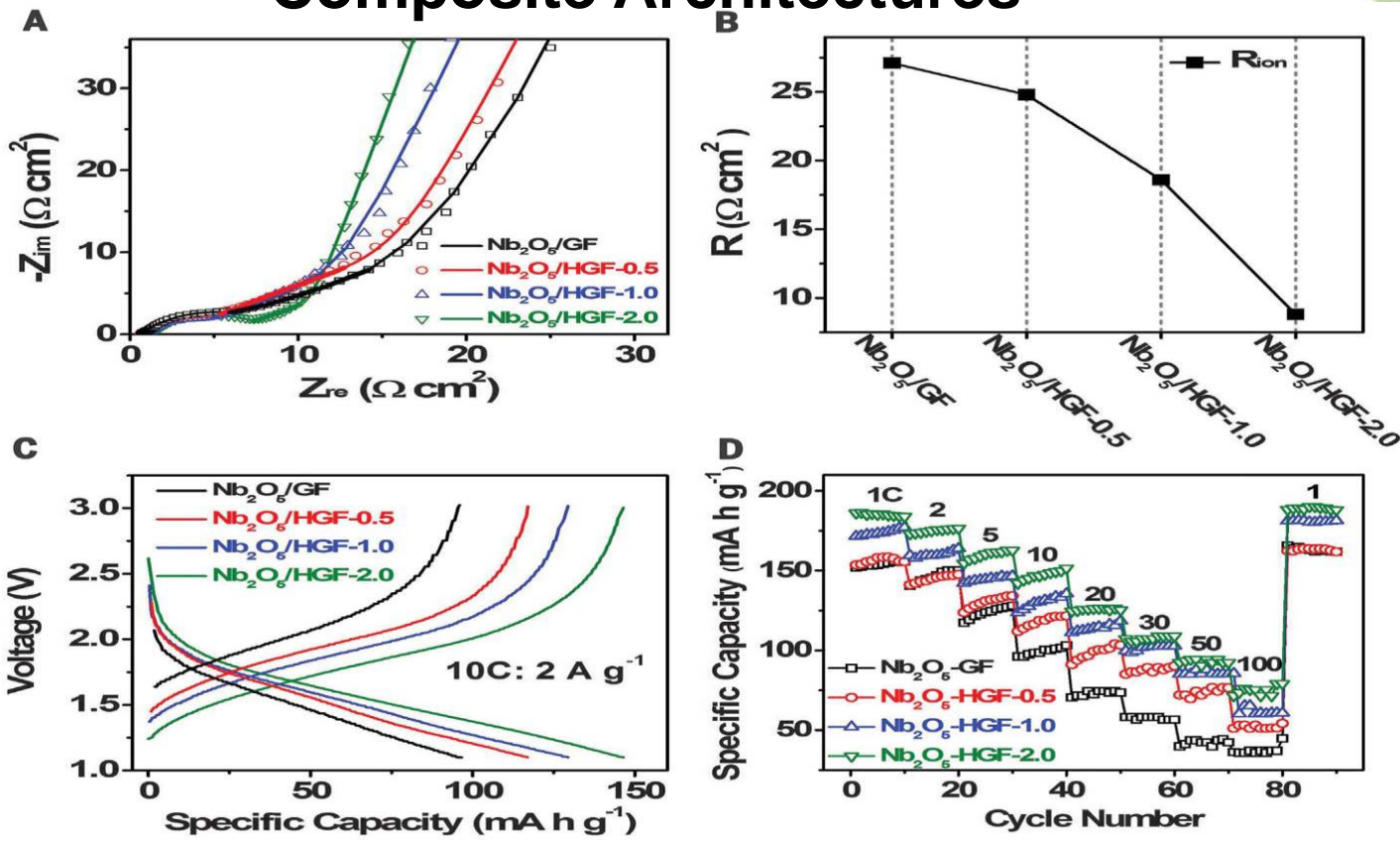


# Three-Dimensional Holey-Graphene/Niobia Composite Architectures



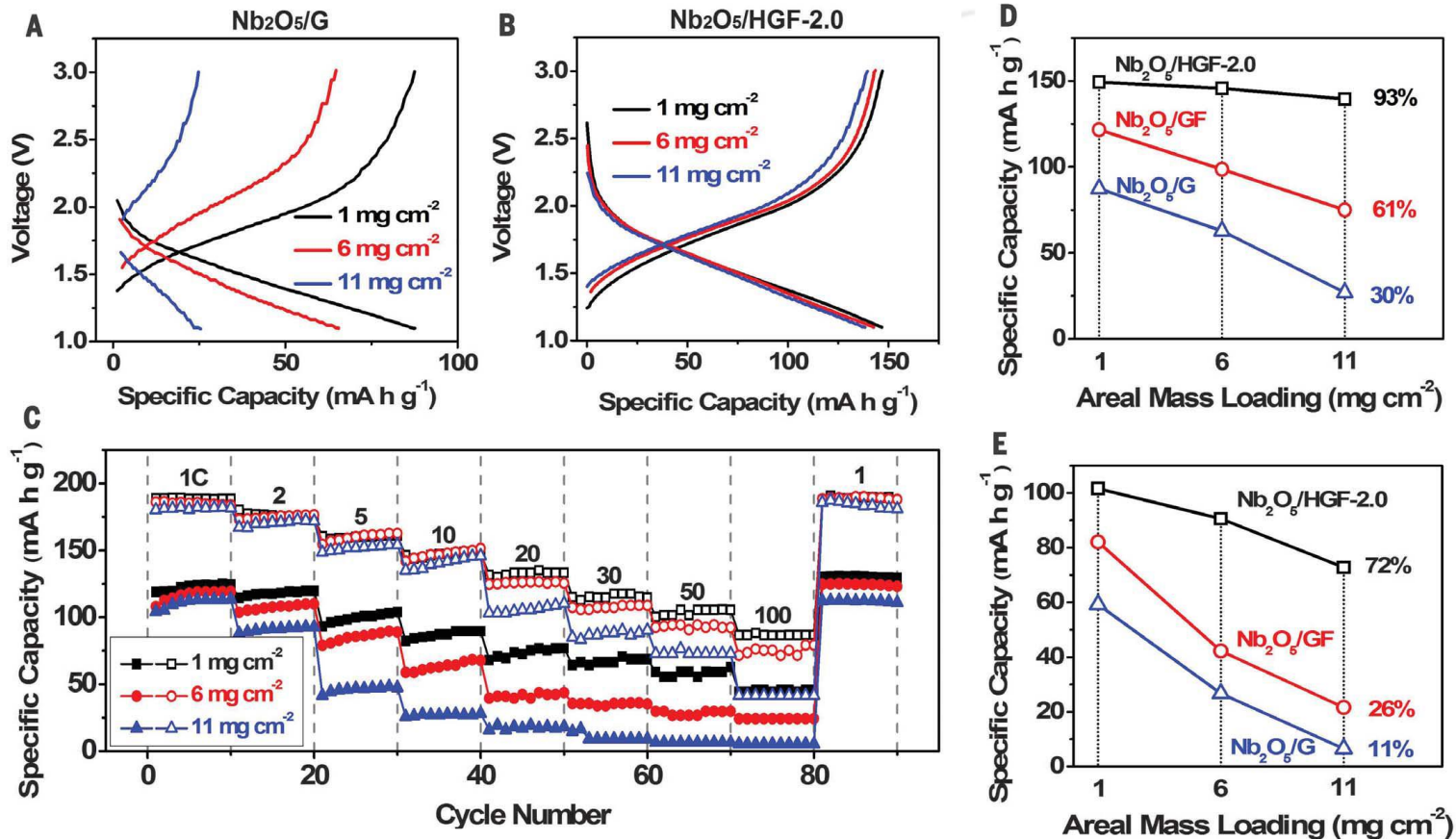
TEM images of graphene sheets with tailored pores obtained by etching in H<sub>2</sub>O<sub>2</sub> for 0, 0.5, 1.0, and 2.0 hours, respectively, (E) Cross-sectional image of Nb<sub>2</sub>O<sub>5</sub>/HGF composite shows 3D hierarchically porous structure, (F) XRD patterns, (G) TEM image of graphene sheets with uniformly decorated T-Nb<sub>2</sub>O<sub>5</sub> nanoparticles, (H) HR-TEM image of T-Nb<sub>2</sub>O<sub>5</sub> nanoparticles, (I) Raman spectra of Nb<sub>2</sub>O<sub>5</sub>/G powder after thermal annealing, and freestanding Nb<sub>2</sub>O<sub>5</sub>/GF and Nb<sub>2</sub>O<sub>5</sub>/HGF electrodes. The D and G bands are characteristic of RGO; The Raman bands at 120, 230, 310, and 690 cm<sup>-1</sup> further confirm the orthorhombic phase of T-Nb<sub>2</sub>O<sub>5</sub>. (J) Comparison of DFT pore size distributions. The prominent pore size shifts from micropores (~1.5 nm) to mesopores (2.7 nm) for the composite prepared from HGO with increasing etching time.

# Three-Dimensional Holey-Graphene/Niobia Composite Architectures



Comparison of Nyquist plots obtained from potentiostatic EIS of a symmetric cell using two identical electrodes (11 mg cm<sup>-2</sup>) at a SOC 0%. The projection of the 45° slope in the high-frequency region is used to determine the ionic resistance for the electrolyte-filled porous architectures in a nonfaradaic process. The open symbols and solid lines represent the experimental and simulation results, respectively. (B) The ionic resistance ( $R_{ion}$ ) as a function of the porosity of the electrode materials. With increasing porosity, the ionic resistance is reduced substantially. (C) Galvanostatic charge-discharge curves of electrodes with tunable nanoporosity at a rate of 10C in the voltage window 1.1 to 3.0 V (versus Li/Li+). (D) Comparison of specific capacities (normalized by the total mass of the electrodes) at various rates (1 to 100C) for composite electrodes with tunable nanoporosity.

# Three-Dimensional Holey-Graphene/Niobia Composite Architectures



Effects of mass loading on electrochemical characteristics. (A and B) Galvanostatic charge-discharge curves for (A) the  $\text{Nb}_2\text{O}_5/\text{G}$  control electrode and (B) the  $\text{Nb}_2\text{O}_5/\text{HGF-2.0}$  electrode at a rate of 10C for the mass loadings of 1, 6, and 11  $\text{mg cm}^{-2}$ . (C) Comparison of the rate performance between 1C and 100C for  $\text{Nb}_2\text{O}_5/\text{HGF-2.0}$  (open) and  $\text{Nb}_2\text{O}_5/\text{G}$  (solid) electrodes under different mass loadings (1, 6, and 11  $\text{mg cm}^{-2}$ ). (D and E) Retention of specific capacity at (D) 10C and (E) 50C as a function of mass loading for three different electrodes. The properties of  $\text{Nb}_2\text{O}_5/\text{HGF-2.0}$  and  $\text{Nb}_2\text{O}_5/\text{GF}$  electrodes were normalized by the total mass of the electrode materials (free of conductive additives and binders); the control  $\text{Nb}_2\text{O}_5/\text{G}$  electrodes were normalized by the total mass of the electrode materials (including binders and conductive additives).

# Conclusions

- The first direction is to integrate active/synergetic nanomaterials
- The second direction is to improve the overall electrical conductivity of metal oxide films/arrays to the largest extent by integrating with diverse conducting agents
- The third direction is the innovation in electrode structure design. For some future applications, good electrodes are expected to be lightweight, full of porosity, mechanically flexible but still robust enough to maintain the power supply.

# Thank you for your attention!



?Any question